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- The structural evolution for the Gr/Co system, as a function of the intercalated Co layer thickness, is herewith investigated;
- the flat-lying adsorption geometry of FePc molecules deposited on Gr/Co substrate, with different degree of corrugation, is determined;
- an interaction mechanism, between the Fe molecule centers and the intercalated Co layer(s), is proposed to be determined by a delicate balance between lateral electric dipoles (due to the rippling of the Gr sheet) and Fe/Co distance.

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Graphene-Mediated Interaction between FePc and Intercalated Cobalt layers

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Abstract

The interaction strength of FePc molecules adsorbed on graphene-Co, obtained by intercalation of Co layer(s) under graphene (Gr) grown on Ir(111), has been investigated by core level photoemission and X-Ray absorption spectroscopy. Upon intercalation of a single layer of Co beneath Gr, the Co atoms are pseudomorphic with the Ir(111) lattice, inducing a rippled Gr layer with valleys and hills regions. FePc molecules, adsorbed on Gr/1 ML Co, are trapped in the valleys regions and the Fe metallic centers strongly interact with the Gr/Co system. The intercalation of further Co layers leads to a relaxation of the Co lattice up to the formation of a commensurate (1×1) Gr/Co system. The Fe centers of the molecules adsorbed on the flat commensurate Gr/Co are found to be less interacting. We have determined that, even though the organic macrocycles of the FePc molecules, flat lying on the Gr/Co system, are completely decoupled by the presence of the Gr intralayer, the Fe-Co interaction is mediated by the graphene spacer and dependent on Gr corrugation.

Keywords: Graphene, Cobalt, Intercalation, Phthalocyanine, Photoemission spectroscopy, Absorption spectroscopy

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1 1. Introduction

2 Magnetic organic molecules are excellent candidates to design novel spin-
3 tronic devices for information processing [1] and storage [2]. High magnetic
4 anisotropy and stable magnetic state of the molecule, up to device work-
5 ing temperature, are mandatory requirements for these purposes. Magnetic
6 substrates, coupled with metal porphyrines and metal phthalocynines (M-
7 Pc), could stabilize the magnetic response up to room temperature [3, 4, 5].
8 However, molecule-metal interactions at the interface can strongly alter the
9 electronic [6, 7]/magnetic [8] properties of the molecules.

10 A Gr sheet, interposed between the molecules and the metal surface, can
11 electronically decouple the adsorbed M-Pcs [9, 10] while increasing their
12 magnetic anisotropy [11]. Ferromagnetic (FM) metals, deposited at room
13 temperature on Gr, can be easily intercalated by thermal annealing at high
14 temperature [12, 13, 14, 15], taking advantage of pre-existing defects in the
15 Gr sheet or via metal-induced defects and healing processes [16]. In par-
16 ticular, intercalation of Co (Fe) has proved to induce a magnetic activity
17 in the Gr sheet [12] as well as tuning the height modulation of the gently-
18 corrugated Gr/Ir moiré superstructure [13, 14, 15]. Highly corrugated moiré
19 superstructures are ideal templates to induce molecule trapping, due to the
20 presence of lateral dipoles/electric fields generated by the local contraction
21 and expansion of the lattice and, therefore, of the density of charge in the Gr
22 sheet [17]. For these systems, a definite interaction between the molecules
23 and the intercalated metal is mediated by the rippled Gr spacer. Recently,
24 we have investigated the FePc adsorption on Gr/Ir(111) intercalated with
25 a single layer of Co, arranged pseudomorphic with the Ir(111) surface [18].
26 The Co layer, in registry with the underlying Ir(111) [13, 19], induces a pro-
27 nounced moiré corrugation of the Gr layer (1.2-1.8 Å for Gr/1 ML Co [13],
28 to be compared with 0.4 Å for Gr/Ir [20]), due to stronger C-Co(Fe) inter-
29 action than C-Ir. The FePc molecules are first trapped into the valleys of the
30 moiré superstructure and a definite interaction of the Fe centers with the un-
31 derlying Co is mediated by the Gr layer [18]. After further intercalation, the
32 Co film structure starts to relax and recover its bulk lattice parameter, with
33 less than 2% [21] lattice mismatch with Gr unit cell, and the Gr/Co interface
34 releases the corrugation and becomes flat. In order to better understand
35 the driving forces of the FePc-Gr-Co interaction, herewith we investigate the

36 FePc adsorption on Gr, as a function of Co intercalation and, accordingly, at
37 decreasing rippling levels, focusing on the flat commensurate Gr/Co system.
38 Core level photoemission and near-edge absorption spectroscopy investiga-
39 tion of the metallic Fe centers unveil a weakening of the interaction strength
40 for FePc molecules adsorbed on Gr in registry with the Co multilayer, when
41 the corrugation is attenuated.

42 2. Experimental Methods

43 Near-edge X-ray absorption fine structure (NEXAFS) measurements were
44 performed at the BOREAS beamline of the Alba synchrotron radiation fa-
45 cility (Barcelona, Spain). The data were collected in the total electron yield
46 (TEY) mode and normalized by the incident flux measured from a gold
47 nanomesh. NEXAFS spectra were acquired in transverse magnetic (TM)
48 and transverse electric (TE) polarizations, by changing the phase of the el-
49 liptically polarizing undulator (EPU), in an experimental geometry analo-
50 gous to Ref. [18] and sketched in the top panel of Fig.4. The core level
51 spectra were acquired at the LoTUS surface physics laboratory in Sapienza
52 University of Rome, with Mg K- α radiation (1256.6 eV) and photoemitted
53 electrons are collected in a normal emission geometry, with a VG Microtech
54 Cam-2 hemispherical analyzer with an overall resolution of 1 eV. The labo-
55 ratories are equipped with analogous ancillary facilities for sample prepa-
56 ration. The procedures adopted for Ir(111) surface cleaning and growth of
57 the Gr layer are fully described in Ref. [18]. Metallic Co, evaporated with
58 a conventional electron beam source, was deposited on the Gr sheet and
59 annealed at 500-800 K for several minutes to favor intercalation of the Co
60 atoms, following Ref. [12, 13, 15]. The Co nominal coverage, determined
61 with a quartz crystal microbalance, was double-checked by evaluating the
62 Ir 4f core level attenuation (at the LoTUS laboratory) or the jump-edge
63 ratio at the Co L_{2,3} absorption edge, pre-calibrated with Auger Electron
64 Spectroscopy at the BOREAS beamline. Commercially available AlphaAe-
65 sar FePc molecules ($\rho = 1.52 \text{ g/cm}^3$) were evaporated with a home-made
66 resistively heated quartz crucible, at a rate of $0.30 \pm 0.05 \text{ \AA/min}$, measured
67 with a quartz crystal microbalance.

68 3. Results and Discussion

69 The LEED patterns of Gr on Ir(111), after the intercalation up to six Co
70 layers, are reported in Figure 1. Graphene, grown on the Ir(111) surface,

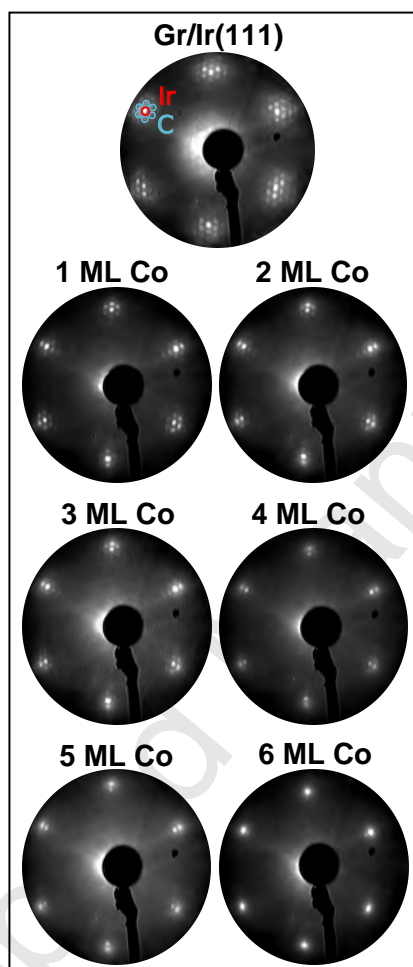


Figure 1: LEED patterns (primary beam energy=140 eV) evolution at increasing number of intercalated Co layers of Gr/Ir(111). A single Co layer is added for each step.

71 exhibits a moiré superstructure due to the lattice mismatch between the Gr
 72 (2.45 Å) and Ir(111) (2.72 Å) lattice parameters. Accordingly, the LEED
 73 pattern (Figure 1, top) presents the main spots of the underlying hexagonal
 74 Ir(111) surface, surrounded by the satellite spots of the C atoms arranged in
 75 a (9.3 × 9.3) moiré superstructure [22]. The moiré superstructure, due to the
 76 corrugated graphene, is preserved and presents the same periodicity and sym-
 77 metry after the intercalation of the first Co layer. Further Co intercalation
 78 induces an attenuation of corrugation along with a progressive and gradual

79 smearing out of the satellite spots up to 4-6 ML of Co, when the mismatch
80 between Gr and Co starts to release and the hexagonal pattern overcomes.
81 Finally, at the completion of the sixth layer, the LEED pattern is a well defined
82 hexagonal (1×1) structure, i.e. the Gr layer is commensurate with the
83 underlying Co substrate [15] and, contrary to previous measurements [23],
84 mostly monodomain. When FePc molecules are adsorbed on Co-intercalated
85 Gr, the symmetry and periodicity of the moiré is unperturbed [18]. The
86 inner and outer spots, characteristic of the superstructure, maintain their
87 distribution and relative distances, thus demonstrating that the adsorption
88 of a single layer of FePc does not modify the Gr moiré periodicity with Co
89 intercalated layer(s), as observed also for FePc on Gr/Ir(111) [24]. Similarly,
90 the (1×1) LEED pattern of flat Gr/6 ML Co is preserved after deposition
91 of a single layer FePc, as reported in the inset of Figure 2.

92 The adsorption geometry of FePc molecules on commensurate Gr/Co can
93 be clarified by polarization-dependent absorption spectroscopy, performed at
94 the N K edge (Figure 2). The spectrum can be divided into two energy re-
95 gions: from 397 to 405 eV, with transitions mainly of π^* character, and
96 above 405 eV, where the σ^* -symmetry transitions dominate the spectrum,
97 in agreement with the existing literature [9, 18, 25]. The dichroic response
98 of the system to linearly (TM and TE) polarized radiation reveals a planar
99 geometry of the FePc molecular units adsorbed on the flat commensurate
100 Gr/Co system, as observed also on both low [9] and highly corrugated [18]
101 Gr sheets. Moreover, the absence of residual signal, due to intermixing of σ
102 and π state in the lower energy region of the spectrum, confirms the decou-
103 pling effect of the graphene layer, avoiding any re-hybridization of the organic
104 macrocycles. Indeed, a molecular distortion of the π^* region of the spectrum
105 was distinctly observed for FePc molecules directly deposited on epitaxial
106 Co(001) films [26], due to a significant mixing of the molecular orbitals lo-
107 cated on the pyrrole macrocycles with the underlying metallic states [26].
108 The presence of the graphene buffer layer in FePc/Gr/Co prevents the cou-
109 pling of the ligand macrocycles with the intercalated Co layer(s) at different
110 degree of corrugation [18], as also observed for FePc/Gr/Ir(111) [9].

111 Although the organic macrocycle is decoupled by the Gr spacer, an in-
112 teraction channel involving the central metal atom was noticed for M-Pc on
113 Gr/Ni(111) [27, 28] and for FePc molecules adsorbed in the valley regions of
114 the Gr/1 ML Co moiré superstructure [18]. In order to better understand
115 the role of graphene corrugation in the interaction process between the Fe
116 centers and the underlying Co, NEXAFS and XPS measurements have been

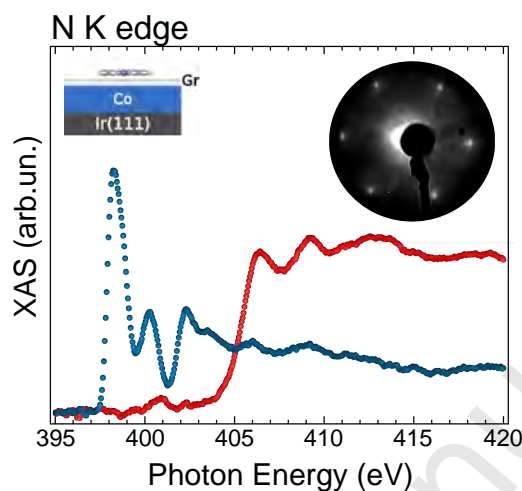


Figure 2: Polarization-dependent, TE (red) and TM (blue), absorption spectra at the N K-edge of 0.4 ML FePc/6 ML Co/Ir(111). Inset: LEED pattern for FePc adsorbed on flat Gr/Co, acquired at liquid nitrogen temperature with a primary beam energy of 140 eV.

117 performed at the Fe L_3 absorption edge and at $2p_{3/2}$ core level of FePc at
 118 different intercalated Co layer thickness.

119 The XPS spectra of the Fe $2p_{3/2}$ core level for FePc molecules adsorbed
 120 on Gr/Co/Ir(111), at increasing thickness of the intercalated Co layer, are
 121 reported in Figure 3, in comparison with molecular adsorption on the gently
 122 corrugated Gr/Ir(111). The broad and asymmetric lineshape of the Fe $2p_{3/2}$
 123 core levels, centered at 709.5 eV, results from the multiplet structure of the
 124 Fe $^{2+}$ central metal ion [6, 29]. In the fitting procedure, so-called partially re-
 125 solved Zeeman splitting, four-fold m_j splitting ($\frac{3}{2}, \frac{1}{2}, -\frac{1}{2}, -\frac{3}{2}$) are accounted
 126 for two non-resolved Gaussian peaks located at 709.5 eV and 711.5 eV bind-
 127 ing energy (BE), whereas a broader satellite is centered at higher BE [6].
 128 The described fitting procedure has been applied to all the spectra reported
 129 in Figure 3. A signature of possible charge transfer and/or orbital inter-
 130 mixing, involving the intercalated Co layer and the central Fe atom of the
 131 molecules, can be revealed by the presence of an extra-component in the Fe
 132 $2p_{3/2}$ core level spectrum [6, 18, 30, 31]. The evolution of the Fe $2p_{3/2}$ core
 133 level lineshape indicates that a charge transfer, from the Gr/Co system to
 134 the central Fe ion, only occurs when the FePc molecules are adsorbed on
 135 the highly corrugated moiré superstructure of Gr/2 ML Co, but not when

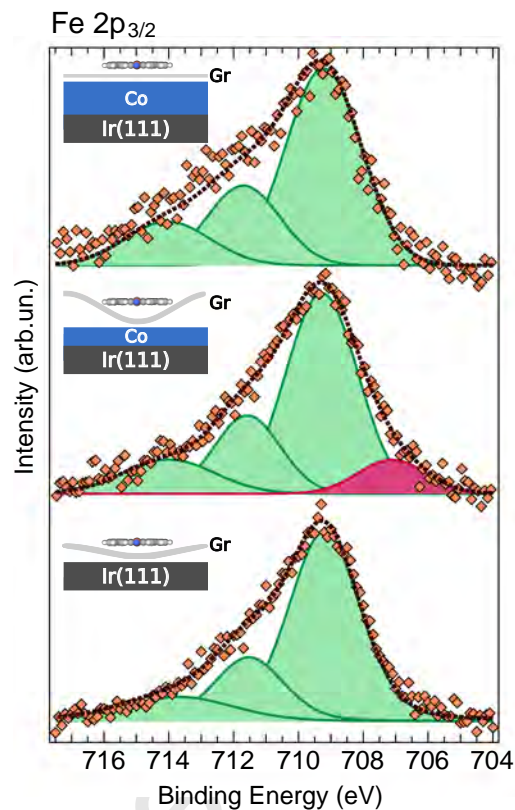


Figure 3: $Fe2p_{3/2}$ core level evolution at increasing intercalated Co layer thickness. In particular, from bottom to top: 1.3 ML FePc on Gr/Ir(111), 1.8 ML FePc on Gr/2.3 ML Co, 1.2 ML FePc on Gr/5.0 ML Co.

136 the molecules are deposited on a gently corrugated moiré (Gr/Ir) or on a
 137 flat commensurate (Gr/Co) system. This evidence confirms recent results on
 138 the Gr-mediated interaction between FePc molecules, adsorbed in the valley
 139 region of the highly corrugated Gr moiré superstructure on the intercalated
 140 single Co layer [18].

141 NEXAFS measurements at the Fe L_3 absorption edge for FePc adsorbed
 142 on Gr/Co/Ir(111) system have been performed to verify the role of Fe metal-
 143 lic centers in the interaction process with the underlying Co layer(s). The
 144 horizontally- (violet) and vertically-polarized (green) light absorption spec-
 145 tra of 0.4 ML FePc adsorbed on flat Gr/6 ML Co and corrugated Gr/1MLCo
 146 are shown in Figure 4, in comparison with a single layer FePc on Gr/Ir(111)

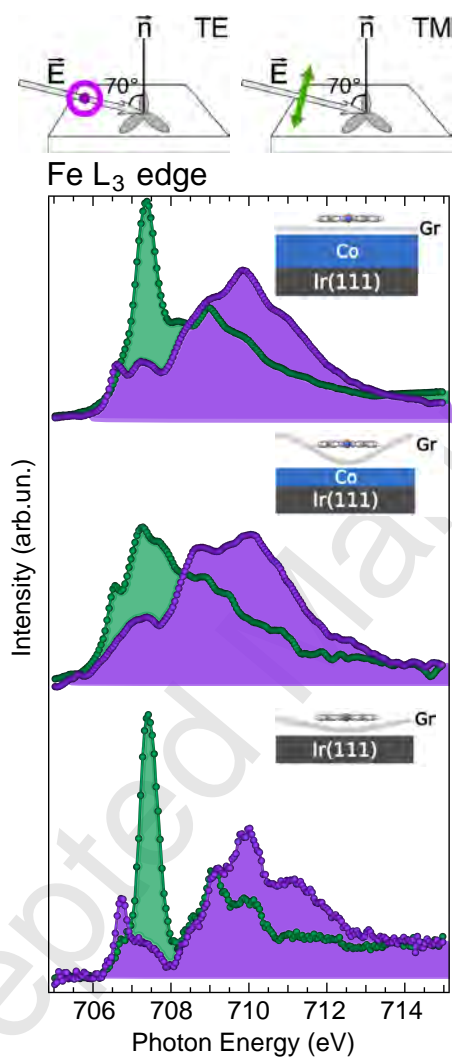


Figure 4: Absorption spectra for TE (violet curves) and TM (green curves) light polarization (see sketch in the top panel for the experimental geometry), acquired at the L_3 edge, for less than 1 ML FePc adsorbed on: Gr/Ir(111) from ref.[9], bottom spectrum; Gr/1 ML Co/Ir(111), central spectrum, and on Gr/6 ML Co/Ir(111), top spectrum.

147 (from Ref.[9]). Indeed, for FePc/Gr/Ir(111) the molecule-metal decoupling
 148 has been proved to be effective and the multiplet distribution of the Fe states
 149 reflects the polarization dependence of the molecular orbitals of a thin FePc
 150 film [32]. The multiplet structure in the Fe L_3 absorption edge of FePc

151 molecules is due to the splitting of the 3d orbitals because of the ligand
 152 field interaction in the D_{4h} symmetry group. Accordingly, thanks to the pla-
 153 nar adsorption geometry, the peaks in the TE and TM polarized absorption
 154 spectra for FePc/Gr/Ir(111) can be ascribed to transitions mainly to d states
 155 perpendicular (e_g , a_{1g}) and parallel (b_{1g}) to the molecular plane, in agree-
 156 ment with a $3E_{1g}$ symmetry ground-state configuration. The most intense
 157 contribution in the TM spectra, located at 707.5 eV, arises from transition
 158 to the out-of-plane a_{1g} state of d_{z^2} symmetry, while the two structures at
 159 706.5 and 709.0 eV are assigned to the two e_g states, with $d_{xz,yz}$ symmetry.
 160 In the TE spectra the in-plane orbitals are probed, and we recognize the e_g
 161 states together with two b_{1g} states, respectively located at 710.0 and 711.3
 162 eV, with a dominant $d_{x^2-y^2}$ character [32, 33, 8]. The intensity of the dif-
 163 ferent contributions in the absorption spectrum for FePc adsorbed on both
 164 highly corrugated and flat Gr sheets, will unravel the interaction strength at
 165 different level of moiré corrugation.

166 A filling of the out-of-plane molecular orbitals is expected when a charge
 167 transfer from the substrate to the molecules occurs, causing a reduction [18],
 168 or even a quenching [8, 33], of the out of plane molecular orbitals. In partic-
 169 ular, while the in-plane states seem less involved in the interaction process,
 170 the out-of-plane molecular orbitals for FePc adsorbed on the valley sites of
 171 Gr/1 ML Co are almost totally quenched, confirming a stronger interac-
 172 tion between the central metal ion of the molecules and the Co, when the
 173 molecules are trapped in the valley regions. On the other side, if the FePc
 174 molecules are deposited on a flat commensurate Gr/Co substrate, the out-
 175 of-plane molecular orbital only partly recover their intensity, indicating that
 176 a Fe-Co intermixing is still present after the relaxation of the moiré cor-
 177 rugation. Though the presence of graphene buffer layer, which completely
 178 decouples the organic macrocycles, the Fe centers are still interacting with
 179 the Co intralayer in all the adsorption configurations here investigated. The
 180 graphene-mediated FePc-Co interaction depends on the Gr rippling and it is
 181 driven by a subtle interplay between the molecular adsorption site induced
 182 by the lateral dipoles in rippled Gr substrate and the distance from the metal
 183 under the graphene spacer [20].

184 4. Conclusions

185 We have investigated the adsorption of FePc molecular units on Co layer(s),
 186 intercalated under a rippled Gr spacer at increasing Co thickness. In agree-

187 ment with previous results, the Co intercalated Gr/Ir surface exhibits a moiré
188 pattern with a decreasing rippling for increasing thickness of the intercalated
189 Co film. The surface potential for molecular adsorption is altered for the
190 intercalated systems compared with the bare G/Ir. In particular, when Co
191 is commensurate with Gr (more than six layers of Co), the FePc are flat
192 lying and the organic macrocycles are decoupled by the Gr layer from the
193 intercalated metal, while a weak interaction of the Fe centers is revealed
194 by the Fe absorption spectra. The strength of Fe centers interaction with
195 the Co increases on highly corrugated moiré superstructures, when few layer
196 of Co are intercalated (less than six Co layers). The interaction with the
197 intercalated Co layers, driven by out-of-plane ligand splitted molecular or-
198 bitals, is mediated by the Gr spacer. The interaction mechanism is a delicate
199 balance between the effect of lateral dipoles of the highly-corrugated moiré,
200 inducing a trapping of the FePc molecules in the valley regions, and the dis-
201 tance of C atoms from the Co template, thereby allowing for locally different
202 electronic coupling. Tuning the graphene rippling can drive the interaction
203 between molecular units and Co magnetic layers, therefore can influence the
204 electronic and magnetic state of the molecular network.

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