

Isostructural phase transition of Fe₂O₃ under laser shock compression

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We present *in-situ* x-ray diffraction and velocity measurements of Fe₂O₃ under laser shock compression at pressures between 38-122 GPa. None of the high-pressure phases reported by static compression studies were observed. Instead, we observed an isostructural phase transition from α -Fe₂O₃ to a new α' -Fe₂O₃ phase at a pressure of 50-62 GPa. The α' -Fe₂O₃ phase differs from α -Fe₂O₃ by an 11% volume drop and a different unit cell compressibility. We further observed a two-wave structure in the velocity profile, which can be related to an intermediate regime where both α and α' phases coexist. Density functional theory calculations with a Hubbard parameter indicate that the observed unit cell volume drop can be associated with a spin transition following a magnetic collapse.

The phase diagrams of iron oxides are notoriously rich with a variety of electronic and structural transitions triggered by pressure or temperature. Due to its relevance to geophysical studies, Fe₂O₃ has been extensively studied under static compression using Laser Heated Diamond Anvil Cells (LH-DAC), up to approximately 113 GPa and 2800 K [1–3]. A series of phase transitions were observed, as well as a possible breakdown of Fe₂O₃ at high temperature into Fe₂₅O₃₂ and Fe₅O₇ [1]. Moreover, a Mott transition and a high-spin to low-spin transition have been evidenced at \sim 50 GPa, although it remains unclear if structural transitions are triggered by the electronic transition or vice versa [4–7]. The α -Fe₂O₃ phase ($R\bar{3}c$) is stable up to 40 GPa. For this phase, a continuous decrease in the c/a ratio is observed with increasing pressure [8–12]. From 40 GPa to 47 GPa, the ι -Fe₂O₃ phase (Rh₂O₃-II type structure, $Pbcn$ orthorhombic) is observed [1, 13]. Above \sim 54 GPa, the ι phase transforms into ζ -Fe₂O₃ phase (a distorted perovskite described in the monoclinic system [8]) stable up to 55 GPa. Above 50-60 GPa, the ζ -Fe₂O₃ phase transforms into η -Fe₂O₃ ($Cmcm$ post-perovskite orthorhombic), while above 67 GPa the possibly metastable θ -Fe₂O₃ ($Aea2$ orthorhombic), is also observed in a limited region between approximately 1000 and 2000 K [1].

Laser-based dynamic compression enables extending *in situ* phase diagram studies to higher pressures than reached by static compression. For that reason, it has

been extensively used to infer mineral properties in planetary and exoplanetary interiors [14–16]. However, the timescales under which phase transformations take place in such experiments are on the order of tens of picoseconds up to microseconds in release, much shorter than the typical planetary timescale. It is thus important to understand the limitations of dynamic compression experiments, in particular when used to inform geophysical studies. Unfortunately, the behaviour of dynamically compressed Fe₂O₃ is poorly understood and experimental results are scarce, largely due to the complexity of target design [17, 18]. Nevertheless, gas gun measurements performed up to 140 GPa [17, 19] show a significant volume drop of \sim 10% at approximately 50 GPa, which may be indicative of the transitions observed statically. Another study reported a significant drop in resistivity at \sim 44-52 GPa using a double-stage light-gas gun on large natural Fe₂O₃ crystals, which can be linked to the Mott transition [18]. However, the crystal structures of Fe₂O₃ under shock compression are unknown, and it remains unclear how these relate to the flurry of phases observed under static compression.

In this letter we present *in-situ* time-resolved x-ray diffraction measurements in shock-compressed Fe₂O₃ using an x-ray Free Electron Laser (XFEL), at pressures between 38-122 GPa. Our results demonstrate that the behaviour of Fe₂O₃ is indeed very different under dynamic compression: we observe none of the high-pressure phases

seen statically, and instead find a single new isostructural phase transition at 50-62 GPa, which we link to a collapse in magnetic ordering.

The experiment was performed at the Matter in Extreme Condition (MEC) end station of the Linac Coherent Light Source (LCLS) [20]. The shock was driven by two synchronized nanosecond lasers (527 nm) incident on the target at an angle of 20° and with a spot size of $300 \mu\text{m}$. We used pulse durations of 5, 10 and 15 ns, with a maximum total energy of 60 J on target. *In-situ* x-ray diffraction (XRD) was performed using the x-ray beam operating in self-amplified spontaneous emission mode, with a photon energy of 7.08 keV, a spectral bandwidth of 25 eV, and pulse duration of 50 fs FWHM [21, 22]. The x-rays were focused onto the target at an angle of 35° with a spot size diameter of $60 \mu\text{m}$, overlapping the focal spot of the optical lasers to probe the planar shock front region. The diffraction signal was measured in transmission geometry on 4 quadruple ePix10k detectors. Azimuthal integration of 2D diffraction images was carried out including solid angle correction, Al filter correction, polarization, and self-attenuation from the target. When this was not possible, integration was performed using the Dioptas software, based on the PyFai library [23, 24], including solid angle correction and polarization. Two Velocity Interferometer System for Any Reflector (VISAR) [25] were used to retrieve the particle velocity and the time when the shock left the Fe_2O_3 sample, and determine the Hugoniot pressure for each data point. VISAR sensitivities were 4.5241 and $1.9890 \text{ km s}^{-1} \text{ fringes}^{-1}$, respectively, with related acquisition time windows of 10 and 20 ns. Two target designs (detailed in SI Section A [26]) were used during the experiment: (1) a $8 \mu\text{m}$ Fe_2O_3 layer sandwiched between a sapphire window and a parylene-N ablator allowing direct measurement of the particle velocity and (2) a $8 \mu\text{m}$ Fe_2O_3 layer on a parylene-N ablator aiming to optimize the XRD signal. Fe_2O_3 was deposited by Physical Vapor Deposition (PVD) and shows a polycrystalline structure of columnar crystallites with preferential orientation (SI section B, C, D [26]).

We show the radially integrated x-ray diffraction patterns of Fe_2O_3 under shock in Fig. 1, measured when nearly the entire Fe_2O_3 layer is shocked, but before shock breakout and release. The pattern recorded prior to the shock at ambient conditions is shown for comparison. At pressures below 42 GPa, the patterns (i) and (ii) show similar features to that of the ambient sample in terms of both peak positions and intensities. Nine α - Fe_2O_3 peaks are identified and shifted toward larger angles due to sample compression. Overall, strong peak broadening is observed in shocked samples, reminiscent of a decrease in crystallite size and an increase in micro-strain effects (lattice parameters fluctuations) under pressure [28, 29]. At intermediate pressures, patterns (iii) and (iv) show four additional peaks, indexed in red on pattern (iii).

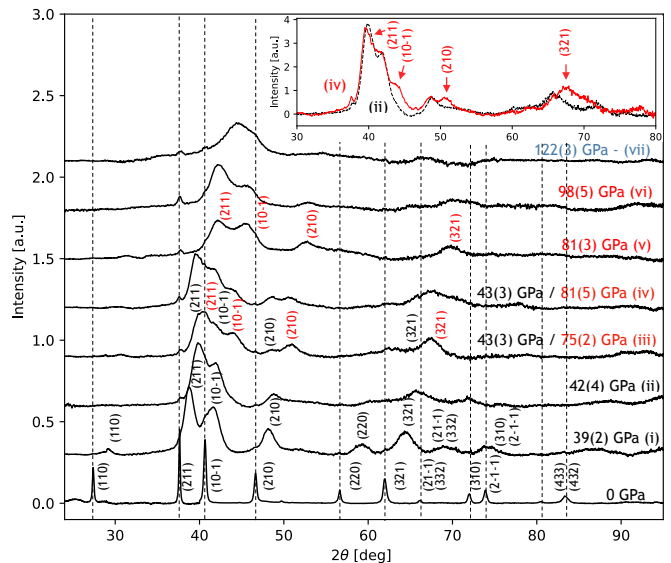


FIG. 1. Radially integrated x-ray diffraction profiles of Fe_2O_3 under shock for targets with a sapphire window between 0-122 GPa. The vertical dotted lines corresponds to the ambient α - Fe_2O_3 peak positions [27]. Miller (hkl) indices are displayed for the low- and high-pressure α structures in black and red, respectively. The inset highlights differences between patterns (ii) and (iv). The two pressures indicated for pattern (iii) and (iv) correspond to the two phases observed in diffraction, and are determined based on VISAR analysis of the double wave structure observed for those data points (fig. 3).

We attribute these to the appearance of a second kind of Fe_2O_3 phase, which we dub the α' phase, having the same crystallographic structure as α but a significantly lower volume. These mixed patterns are thus associated with two pressures, depending on the phase. For pressures above 81 GPa, only four peaks corresponding to the α' - Fe_2O_3 are observed, with a significant shift toward larger angles due to larger compression. At 122 GPa, pattern (vii) shows the disappearance of the diffraction peaks and the appearance of a diffuse signal at scattering angles of 45° ($Q = 2.7 \text{ \AA}^{-1}$) and 55° ($Q = 3.4 \text{ \AA}^{-1}$), which could be interpreted as an amorphous phase (2D diffraction data are shown in Fig. S12 [26]).

It was not possible to perform Rietveld refinements on the data due to the complex microstructure of the samples under shock. Instead, we performed Le Bail refinements, shown in Fig. 2, for data from targets without a sapphire window. At 47(1) GPa a decrease of c/a ratio for the compressed α - Fe_2O_3 phase is observed compared with ambient α - Fe_2O_3 . At 94(2) GPa we identify an α - Fe_2O_3 structure with a significantly smaller volume than the α - Fe_2O_3 observed at 47(1) GPa (225.2 \AA^3 and 265.8 \AA^3 , respectively). In both cases, the α - Fe_2O_3 structure [27] fits the compressed phase well ($R_p = 12.4\%$). The temperature is expected to vary from 400 K at 39 GPa, to 1400 K at 116 GPa, along the Fe_2O_3 Hugoniot.

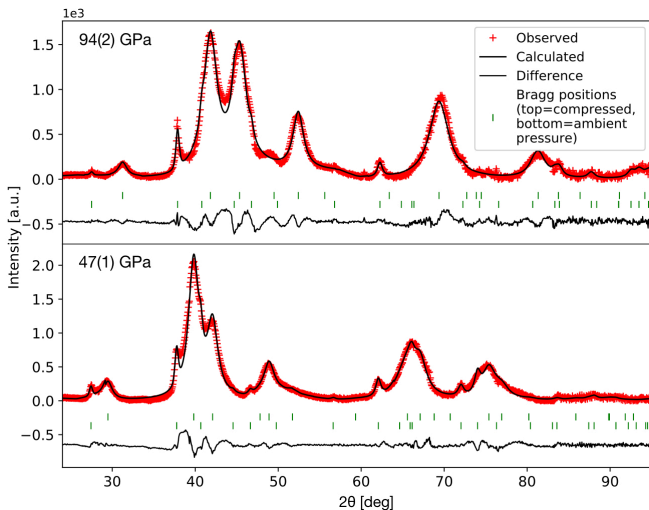


FIG. 2. Le Bail refinements of two radially integrated x-ray diffraction patterns for targets without sapphire window probed just before shock breakout, at pressures of 47(1) GPa and 94(2) GPa. Each pattern was fitted using two α - Fe_2O_3 phases with $R\bar{3}c$ symmetry: one for the α - Fe_2O_3 ambient at 0 GPa (bottom green tick marks) and one for the compressed α - Fe_2O_3 phase (top green tick marks). R_p factors were 10.5% and 12.4% for the 47 GPa and 94 GPa patterns, respectively. Fitted lattice parameters are $a_0 = 5.03$ Å; $c_0 = 13.03$ Å for ambient α - Fe_2O_3 ; $a = 4.87(2)$ Å; $c = 12.94(2)$ Å for 47 GPa; and $a = 4.56(2)$ Å; $c = 12.54(2)$ Å for 94 GPa.

niot based on SESAME 7440 [30]. The Fe_2O_3 Hugoniot thus passes through all stability fields of Fe_2O_3 phases observed under static compression. However, none of the high-pressure phases of Fe_2O_3 reported under static compression were observed (ι - Fe_2O_3 , ζ - Fe_2O_3 , η - Fe_2O_3 , θ - Fe_2O_3 , $\text{Fe}_{25}\text{O}_{32}$, or Fe_5O_7 [1]).

Velocity profiles at the Fe_2O_3 /Sapphire interface for increasing laser intensities are shown as a function of time in Fig. 3. A double-wave structure is systematically observed in VISAR data when the mixing of the α and α' phases is detected in diffraction, as shown for shots in blue. The first wave is labelled P1 and the second wave P2. From the appearance of the α' phase, the P1 wave velocity remains constant at around 1.3 km s^{-1} for all mixed shots regardless of the laser intensity. In contrast, the P2 wave velocity increases with laser intensity and its arrival time becomes shorter. The double-wave structure is also observed for targets without a sapphire window, measured directly from the free surface of Fe_2O_3 (Fig. S11 [26]). This indicates that the double-wave structure is not due to the sapphire window, but is produced within the Fe_2O_3 layer at specific laser intensities. The splitting of the wavefront into two waves is characteristic of a phase transition with a volume change [31, 32]. We thus identify this observation as a signature of the $\alpha \rightarrow \alpha'$ phase transition, which is also observed in diffraction, and associate the P1 and P2 waves with the

α and α' phases, respectively. The pressures in the α and α' phases can then be determined from the P1 and P2 velocities, similarly to previous work on Bismuth [33].

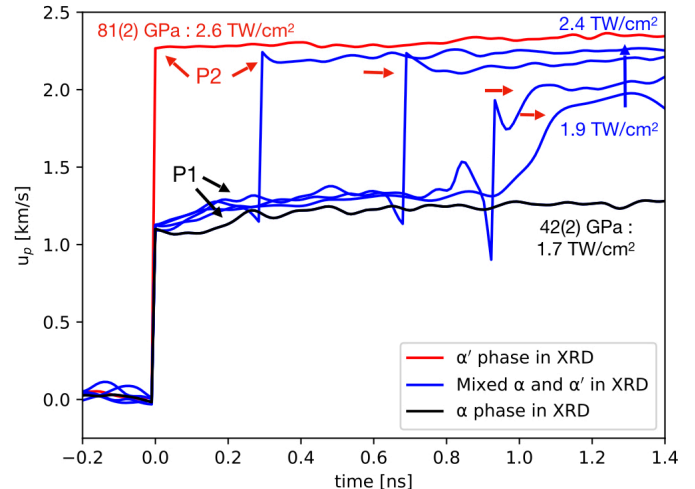


FIG. 3. Particle velocities measured at the rear of the Fe_2O_3 sample for the different pressure regimes. The time origin is set to the shock breakout time in the sapphire. In black we show a low-pressure shock where only a single α phase is observed in x-ray diffraction. The blue curves, taken at intensities between 1.9 and 2.4 TW/cm^2 , correspond to thermodynamic conditions where both α and α' are observed in diffraction. The red curve corresponds to conditions where only a single compressed α' phase is observed. The double-wave structure is indicative of a phase transition.

The unit cell volumes of the α - Fe_2O_3 and α' - Fe_2O_3 phases are plotted against pressure in Fig 4. The α - Fe_2O_3 phase is observed up to 54 ± 2 GPa, and the α' - Fe_2O_3 phase for pressures above 63 ± 2 GPa. We measure a $(11.0 \pm 1.4)\%$ volume drop between the two phases, calculated from x-ray diffraction data showing both in coexistence. This volume drop indicates that the transition from α to α' is a first-order transition.

We acknowledge a pressure gap of 9 ± 3 GPa in our dataset. This could be due to experimental detection limitations, such as 1) the VISAR etalon; 2) the temporal range restricted by the multiple waves in our sample design (see Fig. S6 of the supplementary [26]) that might prevent the detection of the P2 waves with breakout time superior to the temporal range; or 3) the inherent laser-shocked x-ray diffraction peak width, which prevents small α' peaks to be observed. In addition, phase transitions under shock are also subject to temperature effects that might broaden the transition pressure range [31], as well as superheating, kinetic and plasticity effects [34–36] that might affect the onset pressure for α' - Fe_2O_3 synthesis.

Overall, our data yield larger unit cell volumes than previous shock data from gas gun experiments [19] as seen in Fig. 4. However, these previous measurements are intrinsically different to ours, either due to the com-

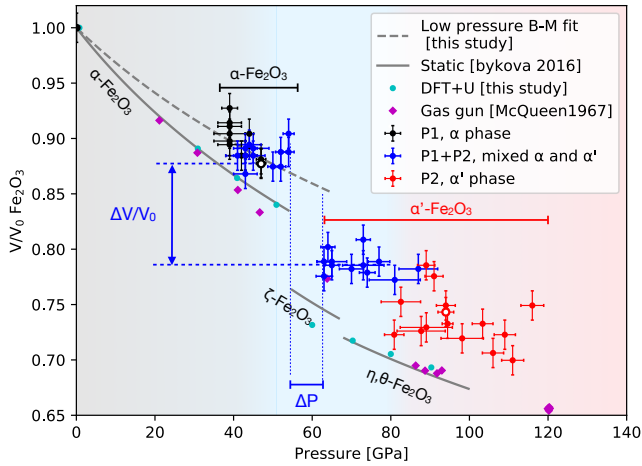


FIG. 4. V/V_0 for α - Fe_2O_3 and α' - Fe_2O_3 determined from x-ray diffraction as a function of pressure, compared with other published work. Points corresponding to the observation of only the α - Fe_2O_3 phase are plotted in black, of only the α' - Fe_2O_3 phase in red, and of the coexistence of both phases in blue. The two empty symbols correspond to data points determined from the Le Bail fits shown in Fig. 2. A transition from α - Fe_2O_3 to α' - Fe_2O_3 phase is observed at 54-62 GPa. It is associated with a relative volume jump $\Delta V/V_0$ equal to 11.0 % and pressure jump of $\Delta P \sim 9$ GPa. A second-order Birch-Murnaghan fit (B-M fit represented with grey dotted line) has been performed for the α phase (see section K of the SI [26]). The relative volume jump between the extrapolated volume for the α phase at 63-65 GPa and the measured volume for the α' phase at the same pressure is 7.7%.

pression strain rate (from 10^5 for gas gun measurement to 10^9 for laser-driven shock compression [37]), the time during which the sample remains under pressure (several hundreds of ns for previous gas gun measurement on Fe_2O_3 [18, 19]), or by the nature of the measurements (optical and macroscopic compared with x-ray bulk and microscopic). In this work, the volume is extracted from the positions of the diffraction peaks. Further, we note that Fe_2O_3 single-crystals are known to behave differently depending of the relative alignment of the compression direction with the crystal axis [18]. Because of how our samples were deposited they do have a notable preferred orientation, and this may well affect the experimental compression values.

Figure 5 shows the evolution of a/a_0 and c/c_0 ratios, where a and c denote the lattice parameters under shock and a_0 and c_0 the parameters at ambient conditions. Both are deduced from Le Bail refinements. Below 54 GPa, the ratios for α - Fe_2O_3 are consistent with static compression data [11], and can be explained by bonding distortion or uneven modifications of the Fe-O-Fe bond lengths [8–12]. This pressure distortion effect is observed in other corundum-structured oxides [8, 38], and can lead to a Mott insulator transition. A discontinuity of 3.6 % is seen between 54-62 GPa for a/a_0 , in

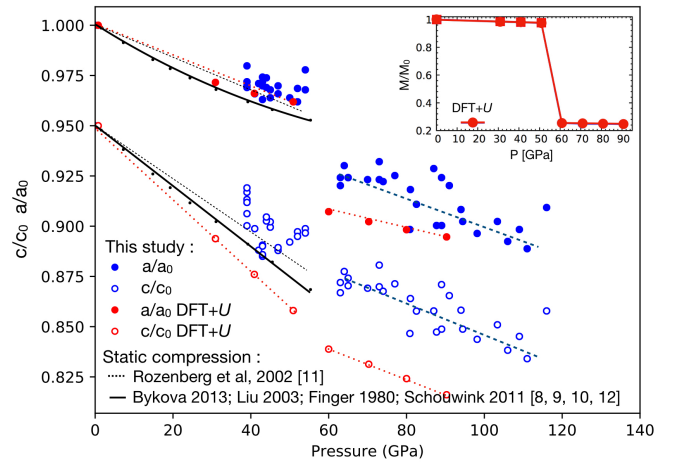


FIG. 5. Lattice parameters for the α and α' phases as a function of pressure. All c/c_0 data are shifted vertically by -0.05 for clarity. Shock data are represented by solid blue dots (a/a_0) and empty blue dots (c/c_0). Dashed blue lines above 60 GPa are drawn to guide the eye. The c/a ratio is shown to vary with pressure for the α phase while it remains constant at 2.73 for the α' phase. Lattice parameters for the α phase from static compression are given as solid black lines for ref. [8–10, 12] and black dotted line for ref. [11]. The results from DFT+ U using a rhombohedral phase are given in red. The inset figure shows the relative magnetic moment of Fe from DFT+ U as a function of pressure. The phase is high spin below 50 GPa, and low spin above 60 GPa.

line with the observed volume drop between the α and α' phases shown in Fig. 4. No discontinuity is observed for c/c_0 . Above 62 GPa we observe a change in the slope of c/c_0 with pressure, while the slope of the corresponding a/a_0 line remains unchanged. This indicates that the c -axis becomes less compressible in the new α' phase, and that the volume drop observed is mostly due to a strong decrease of the a lattice parameter.

As the transition is observed without change of symmetry, the α - Fe_2O_3 to α' - Fe_2O_3 transition could be electronically driven. Two electronic transitions are observed at 50 GPa in static compression experiments: a Mott transition and a simultaneous spin transition [7]. These transitions are not limited by the timescale required for atomic displacement, and can therefore be much faster than a structural transition. Moreover, it has been shown that Mott and spin transitions can lead to a significant volume collapse without structural change [7], as seen, for example, in MnO [39] and FeO [40]. To further investigate such electronic transitions, we performed DFT+ U calculations [41–46] of α - Fe_2O_3 (see SI section L [26] for more detail).

The results of the DFT+ U calculations are shown in Figs. 4 and 5. All main features of interest observed experimentally around the phase transition at 50-60 GPa are reproduced in the simulations: we find a large V/V_0 volume drop ($\sim 10\%$), a discontinuity in the evolution

of a/a_0 but without a change in compressibility, and a change in compressibility but without a discontinuity in c/c_0 . While there remains some discrepancy in the absolute values for the calculated parameters, overall the simulations reproduce the observed trends with good fidelity. The volume drop between 50-60 GPa is correlated to a drop in the local magnetic momentum of iron, as shown in the inset in Fig. 5, decreasing by a factor of 5, i.e. changing from a high to a low spin state. Therefore, we posit that the $\alpha \rightarrow \alpha'$ transition could be explained by a spin transition from a high-spin rhombohedral phase (α -Fe₂O₃) to a low-spin rhombohedral phase (α' -Fe₂O₃ being a low-spin corundum-structured phase). While our DFT+ U calculations are not tailored to identify a Mott transition, it is known from dynamical mean-field theory calculations that the spin transition and the Mott transition are linked and occur simultaneously [47]. Fe₂O₃ undergoes an electronic phase transition independently of the pressure/temperature pathway, being associated with structural phase transition under static compression, but being isostructural upon static decompression [7] and under laser-shock compression. Furthermore, all the Fe atoms are in an octahedral site, favorable to high-spin to low-spin transition upon pressure increase [1, 7]. We thus propose that the $\alpha \rightarrow \alpha'$ transition here observed under laser shock compression is an electronically driven spin transition from high-spin to low-spin, possibly associated with a Mott transition. Importantly, this implies that the electronic transition occurs before any structural transition under shock compression. Previous static compression experiments reported simultaneous electronic (spin and Mott transitions) and structural transitions (to η or θ -Fe₂O₃ phases) [1–7].

To summarize, we have presented *in-situ* time-resolved x-ray diffraction measurements in shock-compressed Fe₂O₃ up to 122 GPa, showing a clear difference in the phase diagram compared with static compression experiments. None of the high pressure phases seen statically were observed dynamically. Instead, we observed an isostructural $\alpha \rightarrow \alpha'$ phase transition around 50-60 GPa, characterized by an 11% volume drop due to a high-spin to low-spin transition, and, possibly, a Mott transition. Our results thus show that the electronic transition(s) observed in Fe₂O₃ under static compression around 50 GPa still occur, but without the associated structural transitions. The effect of the microstructure of our initial sample, uniaxial compression and different pressure-temperature pathways and strain rates involved in static and dynamic compressions [37, 48] should be further explored. This structural phase transition might also require timescales longer than those accessible in our experiment to occur. The structures of the θ and η phases differ significantly from the structure of the α phase: they are composed of FeO₆ prisms and octahedra, whereas the α phase is only composed of FeO₆ octahedra [1]. Our results thus suggest that the fast electronic

transitions (<1 ns) are the ones driving the comparatively slower reconstructive structural transitions to the θ and η phases (>1 ns) [49], and not vice versa. Although a variety of high-spin/low-spin volume jumps have been reported in static compression ([50, 51] and references therein), and inferred from Hugoniot discontinuities under dynamic compression [52–54], here, this volume jump is observed in-situ together with crystallographic measurements. It is the first time that an electronic transition is reported under dynamic compression, without the associated structural transition observed under static compression. Differences have already been reported between static and dynamic phase diagrams such as pressure lowering [34, 55], absence of phase transition [56] or shock amorphization [57]. Our results raise questions on the electronic phase transition mechanisms and kinetics occurring under shock compression, and highlight the need for further investigation.

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