

# Mechanical–electrochemical coupling theory of bacterial cells

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## Abstract

The environmental adaption, growth, motion, and other dynamic behaviors of bacteria are closely associated with the coupled mechanical–electrochemical properties of their subcellular structures, but the underlying regulatory mechanisms remain unclear. The mechanical responses of bacteria are difficult to be elucidated by the traditional models without considering the mechanical, chemical, and electric coupling effects. In this paper, a mechanical–electrochemical theory is constructed to investigate the deformation behaviors of bacterial cells. A bacterium is treated as a bilayer structure consisting of a negatively charged polysaccharide capsule and an elastic envelope subjected to the turgor pressure. This model is used to reveal the regulating roles of the electrostatic double-layer force and osmosis under different electrolyte conditions. A good agreement is found between the theoretical predictions and the experimentally observed three-stage nanoindentation responses of *Klebsiella pneumoniae* bacterial cells. Furthermore, we investigate the mechanical–electrochemical coupling mechanisms in the compression resistance of the bacterial capsule. The results reveal that the osmosis induced by ionic imbalance and polysaccharide–solvent aggregates plays a significant role in the compression resistance of the capsule. The present model can not only deepen our understanding of the mechanical–electrochemical coupling mechanisms of bacterial cells at the subcellular scale, but also holds promise for applications in characterizing their mechanical properties.

**Keywords:** Bacteria; Mechanical–electrochemical coupling; Nanoindentation; Osmosis; Electrostatic double-layer force

## 1 **1. Introduction**

2 Bacteria have strong adaptabilities to complex physiological environments, such  
3 as temperature, osmosis, light, and antibiotics (Ohyama et al., 1992; Pilizota and  
4 Shaevitz, 2012; Rojas et al., 2014). The mechanical properties of bacterial cells are  
5 crucial in their adaptabilities and dynamic behaviors, e.g., growth, division, and motion  
6 (Jiang and Sun, 2010; Liu et al., 2020; Monteiro et al., 2015). The capsular  
7 electrochemistry, envelope elasticity, and cytoplasmic turgor pressure are strongly  
8 coupled in the mechanical responses of bacteria to external stimuli. The turgor pressure  
9 stems from the osmotic difference between the cytoplasm and extracellular  
10 environment, and points toward the cell envelope. The capsule on the bacterial surface  
11 primarily consists of a polysaccharide molecule (PM) network, which carries negative  
12 charges and imbibes water (Gaboriaud et al., 2008; Moxon and Kroll, 1990; Vadillo-  
13 Rodriguez et al., 2004). The negatively charged and hydrated polymeric network  
14 determines the compression resistance of the capsule (Mei et al., 2006), thus having a  
15 remarkable influence on the overall mechanical properties of a cell when subjected to  
16 external loads. The cell envelope, which is remarkably stiff and confers a high tensile  
17 strength of bacteria cells, maintains the integrity of cell morphology (Arnoldi et al.,  
18 2000; Boulbitch et al., 2000; Deng et al., 2011). Both the polysaccharide capsule and  
19 the cell envelope pose physical barriers to the external environment. To construct a  
20 mechanical model of bacterial cells, we need to consider not only the coupling  
21 mechanism among the mechanical–electrochemical cues but also their layered  
22 structures consisting of a negatively charged polysaccharide capsule and an elastic  
23 envelope.

24 Considerable experimental and theoretical efforts have been directed toward  
25 revealing the contributions of mechanics and electrochemistry of subcellular structures  
26 to the overall properties of bacteria (Gaboriaud et al., 2008; Mei et al., 2006; Phanphak  
27 et al., 2019; Wang et al., 2013). Atomic force microscope (AFM) provides a powerful  
28 tool to investigate the elasticity, viscoelasticity, and electrostatic interaction of

individual bacterial cells at the nanoscale level and under various (near-) physiological conditions (Camesano and Logan, 2000; Pelling et al., 2005). Previous theoretical studies have treated the polysaccharide capsule as a brush-like structure and used the Pincus theory to describe the compression resistance, which is mainly determined by the polymer density of the polyelectrolyte brush layer (Abraham et al., 2000; Considine et al., 2001; Wang et al., 2013). However, these previous models are difficult to account for the electrochemical behavior of the capsule. Although the Derjaguin–Landau–Verwey–Overbeek (DLVO) theory considers the adhesion, electrostatic, and van der Waals interaction of the charge-regulated bacterial cell surface (Gaboriaud et al., 2008; Hwang et al., 2012), it is hard to be used to treat the electrochemical and mechanical properties of the capsule, which involve the changes in the external osmosis, ionic concentration, and deformation. Hitherto, it remains elusive how these mechanical–electrochemical mechanisms and subcellular structures coherently affect the mechanical properties of bacterial cells.

In this paper, we propose a coupled mechanical–electrochemical model consisting of a negatively charged capsule and an elastic envelope to rationalize the contributions of mechanical and electrochemical properties. Compared with previous theoretical studies of bacterial cells (Arnoldi et al., 2000; Deng et al., 2011), we model a bacterial cell as a bilayer capsule–envelope structure subjected to a turgor pressure, decoding the regulating roles of subcellular mechanical–electrochemical features in its deformation behavior. The polysaccharide capsule is modeled as a charged polymer gel, and the stiff cell envelope is considered as a linear elastic material. To verify this model, the mechanical properties of *Klebsiella pneumoniae* bacteria are probed by AFM nanoindentation. The theoretical predictions agree well with the experimental results. We also find that the osmosis originating from ionic imbalance and polysaccharide–solvent aggregates determines the mechanical resistance of the capsule, which can help elucidate why the mechanical behaviors of bacteria are sensitive to the electrolyte concentration.

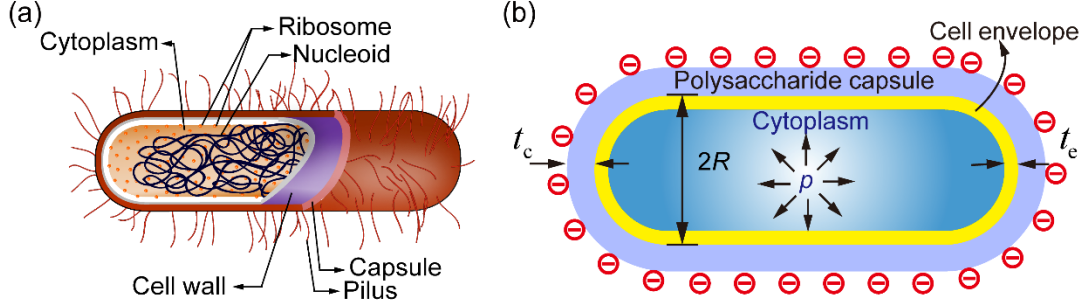
This paper is organized as follows. Section 2 presents a finite-deformation mechanical–electrochemical coupling theory of live bacterial cells. In Section 3, the mechanical properties of *K. pneumonia* bacteria in hypotonic and hypertonic aqueous environments are characterized by AFM-based nanoindentation tests. In Section 4, the theory is applied to analyze the nanomechanical behaviors and compression resistance of a polysaccharide capsule. Finally, we summarize the main conclusions drawn from this study.

## 2. Theory

### 2.1. Layered structure of a bacterial cell

A bacterial cell, such as *Escherichia coli*, *K. pneumonia*, and *Bacillus subtilis*, has a multi-layered structure, mainly consisting of a polysaccharide capsule, a cell envelope, and a fluidic cytoplasm, which have greatly different mechanical properties, as shown in Fig. 1(a). The responses of a bacterium to external stimuli involve the complicated coupling of mechanical, chemical, and electric fields in its solid and liquid phases. In this work, we integrate these coupled mechanisms of multiple subcellular structures in the framework of continuum mechanics. The deformation of a bacterium subjected to external stimuli is considered as an isothermal process at the physiological condition. In the outer layer, the polysaccharide capsule with the PM network is treated as a solid skeleton that ensures the structural and mechanical integrity of the cell, and the interstitial space is filled with fluid, consisting of water and solutes (e.g., chemical factors and oxygen). Thus, we will use the theory of the charged polymeric gel to model the polysaccharide capsule. The cell envelope is simplified as a linear elastic cylindrical shell with two hemi-spherical caps (Vella et al., 2012a, b). The cytoplasm, which is filled with cytosol and organelles, provides a turgor pressure ( $p$ ) acting on the cell envelope. The dielectric polarization of the capsule will be incorporated into the finite deformation theory to describe surface intermolecular interactions. Thus, the bacterium is regarded as a bilayered solid structure consisting of a negatively charged

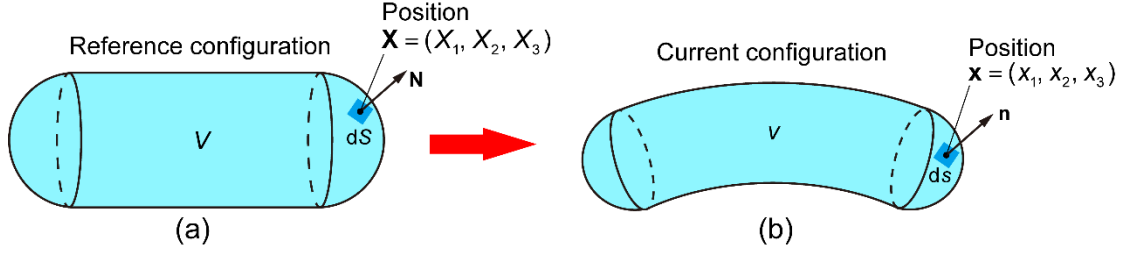
polysaccharide capsule and a cell envelope subjected to the internal turgor pressure, as shown in Fig. 1(b).



**Fig. 1.** (a) Schematics of the multi-layered structure of a bacterial cell. (b) A bilayer capsule–envelope model, including a negatively charged capsule of thickness  $t_c$  and a cell envelope of thickness  $t_e$  subjected to the cytoplasm pressure  $p$ .

## 2.2. Model of polysaccharide capsule

In this work, the polysaccharide capsule is considered as a negatively charged polymeric gel containing a cross-linked PM polymer network as its solid skeleton (Fig. 1(b)). The capsule spacing in the network is filled with fluids consisting of the solvent (i.e., water) and solutes (e.g., ions and antibodies). The mechanical and electrochemical features of the swollen capsule under external stimuli can be characterized by using the hydrogel theory (Hong et al., 2009; Hong et al., 2010). It allows us to mathematically formulate the complicated interplay among the mechanical, chemical, and electric fields. Therefore, in contrast to such previous theories as the Pincus theory that describes the effects of mechanical properties and electrochemistry separately (Hwang et al., 2012; Wang et al., 2013), we consider the coupling mechanisms of these multiple fields, which play an essential role in the deformation behaviors of bacterial cells. For simplicity, we assume here that all states during the considered deformation process are quasi-static and approximately in equilibrium.



**Fig. 2.** Deformation of a polysaccharide capsule in (a) the stress-free reference configuration and (b) the current, deformed configuration.

The stress-free state of a polysaccharide capsule is defined as the reference configuration with total volume  $V$ , surface area  $S$ , and unit outward normal vector  $\mathbf{N}$  at position  $\mathbf{X}$ . Upon loading, the capsule deforms from the reference state to the current configuration with total volume  $v$ , surface area  $s$ , and unit outward normal vector  $\mathbf{n}$  at position  $\mathbf{x}$ , as shown in Fig. 2. Thus, the deformation gradient tensor is  $\mathbf{F} = \partial \mathbf{x} / \partial \mathbf{X}$ .

The PMs in the capsule are highly charged and hydrated (Stokes et al., 2011). Let  $Q(\mathbf{X}, t)$  denote the electric charge density per unit volume, including free electrons, ions, and fixed charges, that is,

$$Q = q + \sum_{\alpha} e z_{\alpha} C_{\alpha} + e z_0 C_0, \quad (1)$$

where  $q$  is the electronic charge density per unit volume of the PM network,  $e$  denotes the elementary charge,  $z_{\alpha}$  is the valence of the fluid phase  $\alpha$ , and  $z_0$  and  $C_0$  are the valence and concentration of the fixed charges, respectively.

The dielectric polarization of the polymeric capsule can be described by Gauss's law of electrostatic fields (Suo et al., 2008). Accordingly, the nominal electric displacement  $\hat{\mathbf{D}}(\mathbf{X}, t)$  in the initial configuration satisfies

$$\nabla_R \cdot \hat{\mathbf{D}} = Q, \quad (2)$$

The deformation of a capsule involves the simultaneous evolutions of the solid PM network and the multiple fluid phases. Thus, the deformation gradient  $\mathbf{F}$ , the nominal concentration  $C_{\alpha}$  of the fluid phase  $\alpha$ , and the nominal electric displacement

1  $\hat{\mathbf{D}}$  are coupled. Let  $w(\mathbf{F}, C_\alpha, \hat{\mathbf{D}})$  denote the Helmholtz free energy density per unit  
 2 reference volume in the capsule. The constitutive equations of the polysaccharide  
 3 capsule can be derived from the energy balance and dissipation inequality (Hong et al.,  
 4 2010; Xue et al., 2017). The first Piola–Kirchhoff stress  $\mathbf{P}$ , the chemical potential  $\mu_\alpha$   
 5 of the  $\alpha$ -th mobile species, and the electric potential  $\Phi$  are given by

$$6 \quad \mathbf{P} = \frac{\partial w}{\partial \mathbf{F}}, \quad (3)$$

$$7 \quad \mu_\alpha = \frac{\partial w}{\partial C_\alpha} + e z_\alpha \Phi, \quad (4)$$

$$8 \quad \nabla_R \Phi = -\frac{\partial w}{\partial \hat{\mathbf{D}}}, \quad (5)$$

9 respectively. Since the electric field is equal to the negative gradient of the potential,  
 10  $\hat{\mathbf{E}}(\mathbf{X}, t) = -\nabla_R \Phi$ , the nominal electric field can be calculated by

$$11 \quad \hat{\mathbf{E}}(\mathbf{X}, t) = \frac{\partial w}{\partial \hat{\mathbf{D}}}. \quad (6)$$

12 In the capsule of a bacterial cell, PMs are entangled together, forming a PM  
 13 network that provides the mechanical and structural integrity, imbibes the solution, and  
 14 carries fixed negative charges (Hammerschmidt et al., 2005). We use a polymeric gel  
 15 model to describe the prominent mechanical–electrical coupling features of the capsule  
 16 immersed in different electrolyte solutions (Hong et al., 2009; Hong et al., 2010;  
 17 Marcombe et al., 2010). Since PMs suspended in a solution of high dielectric constant  
 18 are usually charged (Israelachvili, 2015; Wang et al., 2013), we consider the dielectric  
 19 polarization following the Flory–Rehner theory. Thus, the Helmholtz free energy  
 20 density of the capsule contains four main parts,

$$21 \quad w = w_n + w_s + w_i + w_p, \quad (7)$$

22 where  $w_n$ ,  $w_s$ ,  $w_i$ , and  $w_p$  are the energies resulting from the deformation of the  
 23 PM network, the mixing of the polymers and the solvent, the mixing of the solvent and  
 24 ions, and the polarization of the capsule, respectively.

25 According to the Flory–Rehner theory (Flory, 1953), the capsule can be regarded



as an entropy elastomer, the free energy density induced by the PM network stretch can be expressed as

$$w_n = \frac{1}{2} N k_B T (\mathbf{F} : \mathbf{F} - 3 - 2 \ln J), \quad (8)$$

where  $N$  is the number of polysaccharide chains per unit volume in the dry network, and  $k_B$  is the Boltzmann constant. The conformational entropy of chains obeys the Gaussian statistics. The volume ratio induced by capsule swelling is usually very large, that is,  $J \gg 1$ .

The free energy density induced by mixing the polysaccharide and the solvent is expressed as (Flory, 1942; Huggins, 1942)

$$w_s = k_B T C_w \left( \ln \frac{C_w \nu_w}{1 + C_w \nu_w} + \frac{\zeta}{1 + C_w \nu_w} \right), \quad (9)$$

where  $C_w$  is the concentration of the solvent in the initial configuration,  $\nu_w$  the volume per solvent molecule, and  $\zeta$  a dimensionless parameter characterizing the enthalpy of mixing, respectively. The free energy  $w_s$  is the contribution from the entropy and the enthalpy of mixing the PM and the solvent.

The free energy  $w_i$  induced by mixing the solvent (i.e., water molecules) and the solute ions depends on the entropy of mixing (Hong et al., 2010). It takes the form

$$w_i = k_B T \sum_m C_m \left( \ln \frac{C_m}{C_m^{\text{ref}} C_w \nu_w} - 1 \right), \quad (10)$$

where  $C_m$  is the nominal concentration of the solute  $m$ , and  $C_m^{\text{ref}}$  is the reference concentration of the solute  $m$ .

According to Zhao et al. (2007), we assume that the capsule is an ideal dielectric elastomer with the free energy of polarization:

$$w_p = \frac{1}{2\epsilon} \mathbf{D} : \mathbf{D}, \quad (11)$$

where  $\epsilon$  is the permittivity of the capsule, and  $\mathbf{D}$  is the true electric displacement in the current configuration, and  $\mathbf{D} = J^{-1} \mathbf{F} \cdot \hat{\mathbf{D}}$ . The permittivity  $\epsilon$  is estimated by the

1 volumetric average of the solvent and PM network:

$$2 \quad \varepsilon = \frac{\varepsilon_n + v_w C_w \varepsilon_w}{1 + v_w C_w}, \quad (12)$$

3 where  $\varepsilon_n$  and  $\varepsilon_w$  are the permittivity of the network and solvent, respectively.

4 In general, the volume change of the PMs in the capsule is negligible in  
5 comparison with that induced by swelling. Thus the constraint condition can be given  
6 by

$$7 \quad 1 + \sum_{\alpha} C_{\alpha} v_{\alpha} = J, \quad (13)$$

8 where  $v_{\alpha}$  is the volume per phase  $\alpha$ .

9 Using the constraint condition in Eq. (13), the Helmholtz free energy density  $w$   
10 can be recast as

$$11 \quad w = w_n + w_s + w_i + w_p + \Lambda \left( 1 + \sum_{\alpha} v_{\alpha} C_{\alpha} - J \right), \quad (14)$$

12 where  $\Lambda$  is the Lagrange multiplier.  $\Lambda$  can be interpreted as the overall fluid pressure  
13 in the capsule, expressed as  $\Lambda = \Pi + p$ , where  $\Pi$  is the osmotic pressure, and  $p$  is  
14 the fluid pressure in the external solution.

15 Substituting Eq. (14) into (3) and using Piola transformation, the Cauchy stress is  
16 derived as

$$17 \quad \boldsymbol{\sigma} = J^{-1} \mathbf{F} \cdot \mathbf{P} = \frac{Nk_B T}{J} (\mathbf{F} \cdot \mathbf{F}^T - \mathbf{I}) + \frac{1}{2\varepsilon} (2\mathbf{D} \cdot \mathbf{D}^T - \mathbf{D}^T \cdot \mathbf{D} \cdot \mathbf{I}) - \Lambda \mathbf{I}, \quad (15)$$

18 where the first term on the right-hand side is the contribution due to the network's  
19 entropy, and the second term is the Maxwell stress depicting the effect of the electric  
20 field on the deformation.

21 Substituting Eq. (14) into (4), the electrochemical potentials of the solvent and the  
22 solute  $m$  are obtained respectively as

$$23 \quad \mu_w = k_B T \left[ \ln \left( \frac{v_w C_m}{1 + v_w C_w} \right) + \frac{1}{1 + v_w C_w} + \frac{\zeta}{(1 + v_w C_w)^2} - \sum_m \frac{C_m}{C_w} \right] + \Lambda v_w, \quad (16)$$

$$\mu_m = e z_m \Phi + k_B T \ln \left( \frac{C_m}{v_w C_w C_m^{\text{ref}}} \right) + \Lambda v_m. \quad (17)$$

Substituting Eqs. (14) into (6) and using Piola transformation, the true electric field in the current configuration (Zhao et al., 2007) is derived as

$$\mathbf{E} = \hat{\mathbf{E}} \cdot \mathbf{F}^{-1} = \frac{\mathbf{D}}{\varepsilon}. \quad (18)$$

### 2.3. Constitutive relation of cell envelope

In general, the bacterial envelope can be considered as a pressurized elastic shell subjected to a turgor pressure. Previous work has shown that the elastic modulus of the cell envelope is linearly related to the turgor pressure (Deng et al., 2011). Under a fixed internal pressure, the elastic modulus is regarded as a constant, as assumed in the previous work (Arnoldi et al., 2000; Vella et al., 2012a, b). Thus, the linear elastic model is adopted to describe the mechanical response of the cell envelope. Its constitutive relation is

$$\boldsymbol{\sigma} = \frac{E_e}{1+\nu} \boldsymbol{\varepsilon} + \frac{E_e \nu}{(1+\nu)(1-2\nu)} (\text{tr} \boldsymbol{\varepsilon}) \mathbf{I}, \quad (19)$$

where  $E_e$  is the Young's modulus,  $\nu$  the Poisson's ratio, and  $\mathbf{I}$  the identity tensor.

### 2.4 Summary of the theoretical model

In summary, we have developed a mechanical–electrochemical coupling model to investigate the response of a bacterial cell to external stimuli. The cell is considered as a bilayer capsule–envelope structure subjected to the turgor pressure. The polysaccharide capsule layer is treated as a negatively charged polymeric gel with the coupling mechanisms of mechanical, chemical, and electric cues, as described by the constitutive relations in Eqs. (15)–(18). The cell envelope layer is treated as a linear elastic shell. Through the combination of the mechanical equilibrium equation and specific boundary conditions, we can solve the field variables of stresses, fluid chemical potentials, solute concentrations, and electric potentials. In what follows, this theory

will be used to analyze the nanoindentation tests and the compression resistance of a bacterial cell under different electrolyte conditions.

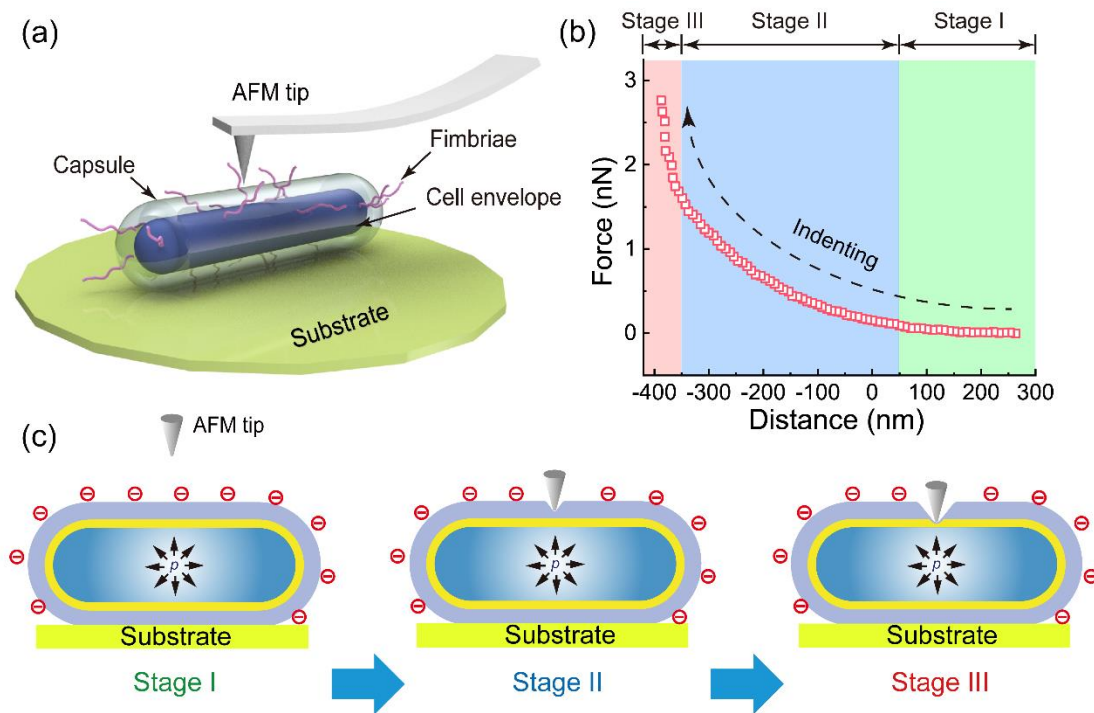
### 3. Nanoindentation tests of bacterial cells

*K. pneumonia* strain is a rod-shape Gram-negative pathogen that may cause a variety of hospital-acquired infections, e.g., surgical wound infections (Jenney et al., 2006). The subcellular structure of wild-type *K. pneumonia* is the same as that described in Section 2.2. We chose this strain as an example to explore the effects of osmosis and electrochemistry on the deformation of bacterial cells. A series of AFM nanoindentation tests of *K. pneumoniae* cells submerged in Milli-Q water and 100 mM  $\text{CaCl}_2$  were conducted, as illustrated in Fig. 3(a). The two electrolyte solutions correspond to hypertonic and hypotonic conditions, respectively.

The radii of the bacterial cells were measured from the cross-sectional analysis of the AFM contact-mode three-dimensional height images (Zhang et al., 2021). The measurements show that the wild-type bacterial cells in Milli-Q water and 100 mM  $\text{CaCl}_2$  have average radii of  $352 \pm 31$  nm and  $356 \pm 38$  nm, respectively, which have a small difference of only 1.8% (Wang et al., 2013). Previous experiments found that, after osmotic shock, the response of a bacterial cell relies on its mechanosensitive channels, which allow water and cytoplasmic solutes to pass across the cell membrane, helping to maintain the relative stability of the difference between the internal and external osmotic pressures, also termed as the turgor pressure. Through this mechanism, the volume of a bacterial cell can recover to its initial value in several minutes (Buda et al., 2016; Miermont et al., 2013). This indicates that bacterial cells have an adaptive ability to the variations in the external osmolarity.

It can be seen from our nanoindentation test results in Fig. 3(b) that, for a wild-type bacterial cell in Milli-Q water, the force–distance curve can be divided into three stages. First, Stage I appears before the AFM tip gets in contact with the cell. In the range of  $\sim 300$  nm from the bacterial cell to the negatively charged silicon nitride AFM

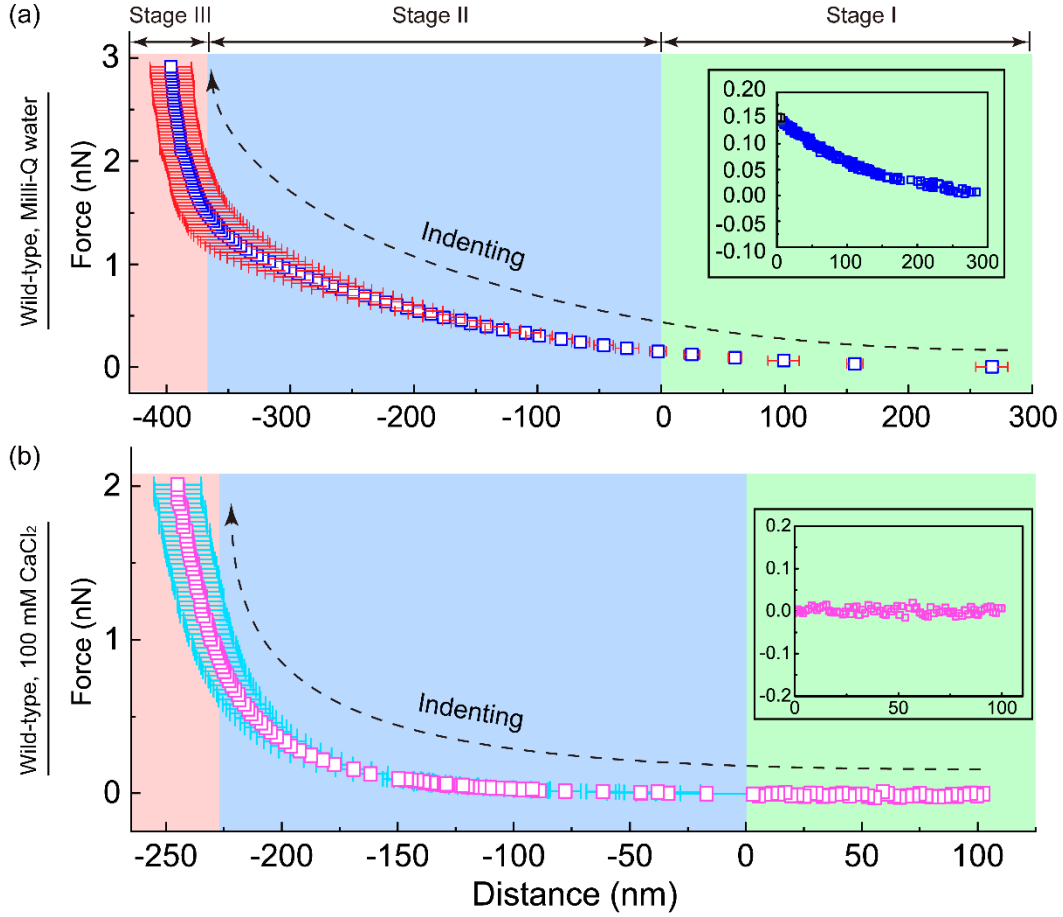
tip, we can measure an increasing repulsive force. This confirms the previous electrohydrodynamic study that the polysaccharide capsule bears a certain amount of negative charges, which induce electrostatic interaction (Gaboriaud et al., 2008). Secondly, Stage II occurs after the tip has got in contact with and gradually compresses the surface polymer layer of the cell. In this stage, the indentation force–distance curve features a significant nonlinearity. This suggests that the polysaccharide capsule serves as a buffer to bear external stimuli, including the mechanical load and the electrolyte environment. As the indentation depth further increases, the tip presses the cell envelope and the force profile becomes approximately linear, with a much larger slope, which is referred to as Stage III. In this stage, the elastic deformation of the cell envelope plays a dominant role in the indentation response. The entire indentation process is schematically illustrated in Fig. 3(c).



**Fig. 3.** Mechanical response of a wild-type *K. pneumoniae* cell under AFM indentation. (a) Schematic diagram of the indentation test of a bacterial cell with attached fimbriae adhered to a substrate. (b) An experimental nanoindentation force–distance curve of a wild-type (red squares) bacterial cell submerged in Milli-Q water. (c) Schematic

1 diagrams illustrating the mechanisms in the three stages of the nanoindentation  
2 response, in which the resistance of fimbria, also known as pilus, can be ignored due to  
3 its low density (Wang et al., 2013).

4  
5 The indentation responses of a bacterial cell under two different osmotic  
6 conditions are compared in [Fig. 4](#), which are closely associated with the influences of  
7 osmosis and electrochemistry on the cell deformation. Although the force–distance  
8 curves in Stage II and III of 100 mM  $\text{CaCl}_2$  are similar to those in Milli-Q water, they  
9 have several differences as follows. First, for a wild-type bacterial cell in 100 mM  $\text{CaCl}_2$ ,  
10 the electrostatic repulsive force is almost negligible in Stage I. This is because the  
11 negative charges on the PMs are neutralized by the  $\text{Ca}^{2+}$  ions in the solution. Second,  
12 the nonlinear Stage II of the cell in  $\text{CaCl}_2$  is relatively shorter than that in Milli-Q water.  
13 This finding highlights the remarkable change in the structure of the polysaccharide  
14 capsule in response to the added electrolyte (Wang et al., 2013). In the following, we  
15 will use the mechanical-electrochemical coupling theory described in [Section 2](#) to  
16 analyze the indentation behavior of *K. pneumoniae* cells and the compression resistance  
17 of the capsule under different conditions.



**Fig. 4.** Population-averaged indentation curves ( $n \approx 200$  in each osmotic condition) of a wild-type cell in (a) Milli-Q water (blue squares) and (b) 100 mM  $\text{CaCl}_2$  (pink squares). The red and cyan error bars represent the standard deviations under the two conditions. The long-range interaction between the wild-type cells and the AFM tip is shown in the insets of (a) and (b).

#### 4. Theoretical validation and discussions

The above experimental results show that the indentation response of a bacterial cell can be divided into three stages, which stem from both its layered structures and their mechanical and electrochemical properties. In this section, we apply the mechanical–electrochemical coupling theory described in [Section 2](#) to decode the underlying mechanisms of the three-stage nanoindentation response of a bacterial cell. The effect of external electrolytes on the compression resistance of the capsule will be examined to show the application of the theory. All parameters used in the present

model, and their resources, are listed in Table 1.

We first validate the mechanical-electrochemical coupling theory proposed in Section 2 by comparing it with the results of the experiments (Fig. 4) and finite element (FEM) simulations. To track the three-stage nanoindentation responses, we implement the proposed theory into the commercial software *Abaqus 6.14*. The approach of user-defined FEM simulation of the mechanical–electrochemical coupling model is given in Appendix I.

**Table 1.** Parameters used in the calculation.

Symbol	Description	Value	Source
$k_B$	Boltzmann constant	$1.38 \times 10^{-23} \text{ J}$	
$T$	Absolute temperature	298 K	
$v_w$	Volume of a water molecule	$3 \times 10^{-29} \text{ m}^3$	
$\omega_t$	Surface charge density on AFM tip	$-0.032 \text{ C m}^{-2}$	Mueller et al. (1999)
$\zeta$	Dimensionless parameter of enthalpy of mixing	0.2	Xue et al. (2017)
$z_0$	Charge of a PM	−4	Xue et al. (2017)
$C_0$	Concentration of fixed charges	$3.3 \times 10^{25} \text{ m}^{-3}$	Gaboriaud et al. (2008)
$t_e$	Thickness of cell envelope	26 nm	Mularski et al. (2017)
$R_t$	Curvature radius of AFM tip	20 nm	Our experiments
$\theta$	Cone angle of AFM tip	15°	Our experiments
$\omega_c$	Surface charge density on the capsule in Milli-Q water	$-2 \times 10^{-4} \text{ C m}^{-2}$	Our experiments
$N$	Number density of polysaccharide chains	$2.6 \times 10^{23} \text{ m}^{-3}$	Our experiments
$R$	Bacterial radius	350 nm	Our experiments
$P$	Turgor pressure	80 kPa	Our experiments
$E_e$	Young's modulus of cell envelope	115 MPa	Our experiments



#### 4.1. *Electrostatic interaction*

Due to the presence of negative charges on the capsule layer, an electrostatic repulsion occurs between the bacterial cell and the AFM tip. The electrostatic interaction plays an indispensable role in bacterial adhesion (Guo et al., 2018; Hayashi et al., 2001; James et al., 2017; Poortinga et al., 2002; Straub et al., 2019). In Stages II and III of the nanoindentation curve in Fig. 4, the electrostatic repulsive interaction force is much lower than the contact force, and thus the contribution of electrostatics to the deformation of the capsule is insignificant, as shown in Fig. 4(a). Therefore, the electrostatic interaction is not considered in the numerical model when simulating the indentation, instead, it is superimposed on the numerically obtained indentation force–depth curve. In what follows, we will first determine the theoretical relation between the electrostatic force and the external electrolyte, and then simulate the two typical experimental situations shown in Fig. 4 based on this relation of electrostatic repulsion and the mechanical-electrochemical coupling theory described in Section 2.

The electrostatic repulsive force between the indenter and the cell depends on the electrochemical property of the polysaccharide polymer layer and is regulated by the electrolyte environment. It originates from a double electric layer with a characteristic thickness, referred to as the Debye length ( $\lambda_D$ ), near the interface between the capsule and the external solution. When the negatively charged AFM tip approaches the capsule, it is subjected to an increasing repulsive electrostatic force (also called as the double-layer force) induced by the double electric layer (Gaboriaud et al., 2008; Mueller et al., 1999; Rijnaarts et al., 1999). The Debye length  $\lambda_D$  for Milli-Q water is  $\sim 100$  nm (Carambassis et al., 1998), whereas in the 100 mM  $\text{CaCl}_2$  solution, it is as small as  $\sim 0.55$  nm (Israelachvili, 2015). Due to the nature of electrostatic interaction, the double-layer force is obvious in Milli-Q water but negligible in  $\text{CaCl}_2$  solution. This helps understand the difference in the nanoindentation responses in Stage I under diverse electrolyte conditions in our experimental results, as shown in Fig. 4.

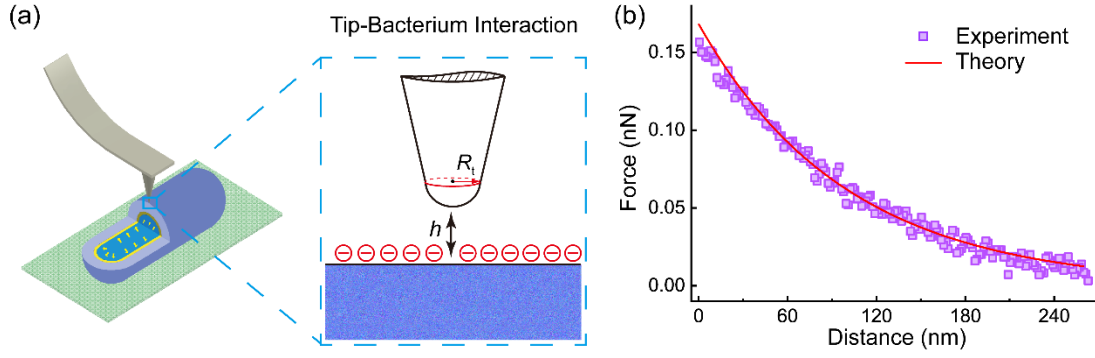
The electrostatic force between the AFM tip and the bacterial cell depends on the

1 ionic concentrations and the surface charge densities on the bacterial surface and the  
 2 AFM tip. Since the bacterial size is much larger than the probe, we simplify the  
 3 electrostatic repulsion as an interaction between the small spherical surface of the tip  
 4 and a semi-finite plane (Fig. 5(a)). Applying the Derjaguin approximation, we calculate  
 5 the electrostatic force by (Israelachvili, 2015)

$$F_e = \frac{R_t Z}{\lambda_D} \exp\left(-\frac{h}{\lambda_D}\right), \quad (20)$$

7 where  $R_t$  is the curvature radius of the AFM tip,  $Z = 4\pi\omega_c\omega_t\lambda_D^2/\epsilon$  is the interaction  
 8 constant. Here,  $\omega_c$  and  $\omega_t$  denote the surface charge densities on the capsule and the  
 9 AFM tip, respectively, and  $h$  is the distance between the two surfaces. Therefore, the  
 10 electrostatic force decays exponentially with the distance with a characteristic decay  
 11 length, which is taken as the Debye length,  $\lambda_D$ .

12 It is known from Eq. (20) that the electrostatic force between the indenter and the  
 13 bacterial cell depends on the curvature radius  $R_t$  of the AFM tip, the Debye length  $\lambda_D$ ,  
 14 and the surface charge densities  $\omega_c$  and  $\omega_t$  on the capsule and the AFM tip. In the  
 15 experiments, we adopt an AFM probe with radius of curvature 20 nm and cone angle  
 16 of 15°.  $\lambda_D$  and  $\omega_t$  in Milli-Q water are taken to be 100 nm and  $-0.032 \text{ C m}^{-2}$ ,  
 17 respectively (Mueller et al., 1999; Wang et al., 2013). By fitting the experimental data  
 18 with Eq. (20), the surface charge density on the capsule is found to be  
 19  $\omega_c = -2 \times 10^{-4} \text{ C m}^{-2}$ . As can be seen from the comparison in Fig. 5(b), the theoretical  
 20 result agrees well with the experimental curve.



**Fig. 5.** Electrostatic interaction during the nanoindentation of a bacterial cell. (a) Theoretical model for calculating the electrostatic interaction between the spherical surface of the AFM tip and a semi-infinite plane. (b) Comparison between the theoretical and experimental results for the electrostatic force between the AFM tip and a wild-type bacterial cell in Milli-Q water.

#### 4.2. Numerical simulations of AFM nanoindentation

Now, we simulate the indentation force–depth curves of bacterial cells in the two electrolyte situations shown in Fig. 4 to validate the mechanical-electrochemical coupling theory described in Section 2. In the FEM model, the AFM probe is pressed into the bacterial cell at several points along the cell apex. Based on the transmission electron microscopy images of wild-type *K. pneumoniae* cells (Mularski et al., 2017), the bacterium is modeled as an envelope–capsule bilayer tube. In the contact zone between the bacterial cell and the substrate, the capsule is involved in forming an adhesion layer between the capsule and the substrate (Burks et al., 2003). Thus, we regard the capsule layer in contact with the substrate as a fixed constraint of the displacements along  $x = 0$ . We assume that no relative displacement exists between the capsule and the envelope (no-slip condition), and the connections of different layers are set to be perfectly bonded. The envelope and the capsule were discretized into eight-node plane-strain hybrid elements (CPE8H).

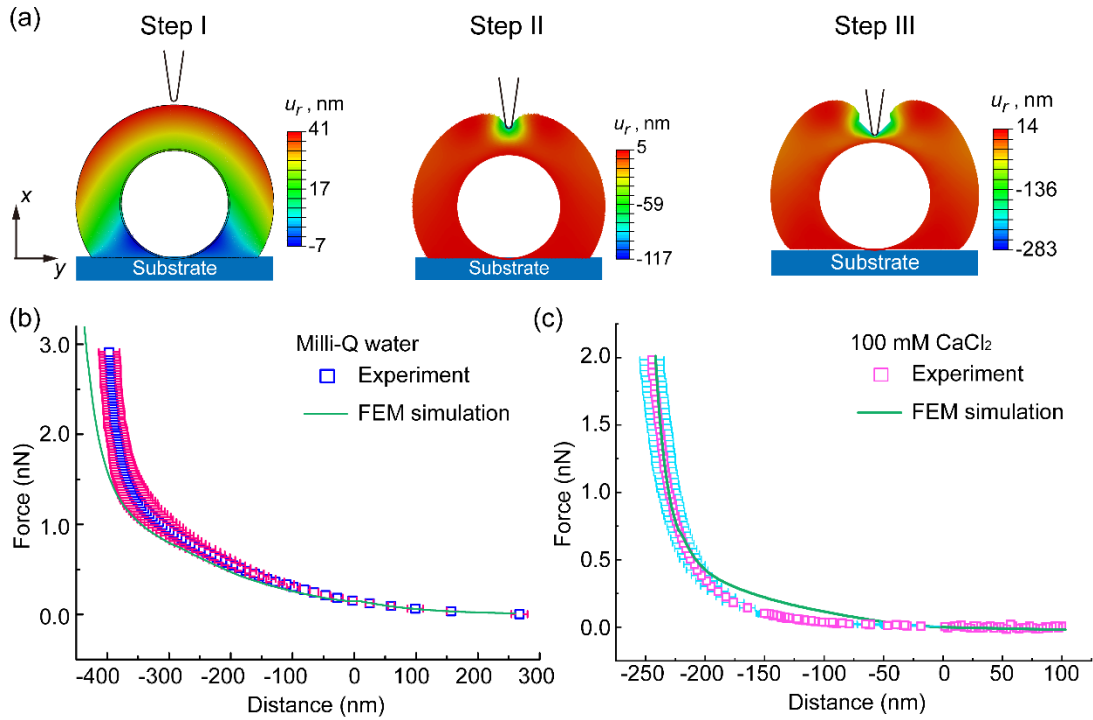
In the simulations, the deformation of the bacterial cell is decomposed into two parts, which are induced by constrained expansion and indentation, respectively. When

the turgor pressure and external electrolyte are added, the cell swells and reaches the state of mechanical equilibrium. As the AFM tip presses into the cell, the capsule and cell envelope deform successively, as a typical example of the FEM simulation shown in Fig. 6(a). The indentation response in Stage II is controlled mainly by the compression resistance of the capsule. Our detailed analysis, see Eq. (27) in Section 4.3, shows that the compressive resistance depends on the ionic concentration  $c_0$  in the external solution, the number density  $N$  of polysaccharide chains, the concentration  $C_0$  of fixed charges, the charge  $z_0$  of a polysaccharide molecule, and the dimensionless parameter  $\zeta$  of the enthalpy of mixing. The values of  $C_0$ ,  $\zeta$  and  $z_0$  are taken according to previous references (Gaboriaud et al., 2008; Xue et al., 2017), and  $c_0$  is obtained in our experiments. However, the polysaccharide chain density  $N$  and the thickness  $t_e$  of the capsule are difficult to measure directly and, thus, estimated from the indentation curves (Fig. 4).

When the indentation enters Stage III, the force–depth curve becomes approximately linear due to the high elastic stiffness of the cell envelope and the turgor pressure. From the experimental results, we obtain the indentation stiffness as  $0.074 \text{ N m}^{-1}$  in Milli-Q water and  $0.082 \text{ N m}^{-1}$  in 100 mM  $\text{CaCl}_2$ . In our previous work (Zhang et al., 2021), we proposed a nanoindentation method to simultaneously measure the turgor pressure  $p$  and the Young’s modulus  $E_e$  of the cell envelope of a live bacterium. Using that method, we determine  $p = 80 \text{ kPa}$  and  $E_e = 115 \text{ MPa}$ . Besides, we take the bacterial radius  $R = 350 \text{ nm}$ , and the cell envelope thickness  $t_e = 26 \text{ nm}$ , following Mularski et al. (2017).

In line with our experiments, the normalized concentrations are set as  $v_w c_0 = 1 \times 10^{-6}$  in Milli-Q water and  $v_w c_0 = 2 \times 10^{-3}$  in 100 mM  $\text{CaCl}_2$  solution. In the FEM simulations, the thickness of the capsule is estimated as  $t_e = 320 \text{ nm}$  in the Mill-

Q water and  $t_c=152$  nm in 100 mM  $\text{CaCl}_2$  solution, and the number density of polysaccharide chains is estimated as  $N=2.6\times 10^{23} \text{ m}^{-3}$ . As can be seen from Figs. 6(b) and (c), our numerical simulations based on the proposed mechanical–electrochemical theory are in agreement with the nanoindentation test results. Since the exact values of some parameters used in the simulations, such as the number density  $N$  of polysaccharide chains and the capsule thickness  $t_c$ , are difficult to measure, the simulation results may have small deviations. Due to the individualism of bacterial cells, the bacterial nanoindentation responses also exhibit distinct dispersion. Our experiments evidence that the thickness of the capsule alters in response to the ionic concentration. Besides, Mei et al. (2006) found that the bacterial polysaccharide capsule may collapse into a more densely packed structure through counterion condensation, and thus its compression resistance may alter as the molecular density of the capsule changes. These detailed mechanisms have not been considered in our simulations



**Fig. 6.** FEM simulation for the nanomechanical response of a bacterial cell. (a) A numerical model is established to calculate the three-stage deformation behavior of a

cell under external osmotic situation and AFM indentation, where the normalized concentrations  $v_w c_0 = 1 \times 10^{-4}$  and the capsule thickness  $t_c = 280$  nm. The color represents the radial displacement. Comparison between the numerical and experimental results for the force–depth curves of wild-type bacterial cells (b) in Milli-Q water and (c) in 100 mM  $\text{CaCl}_2$  solution.

#### 4.3. Compression resistance of polysaccharide capsule

As another application of the mechanical–electrochemical theory for the polysaccharide capsule in Section 2, we further analyze the compression resistance of the polysaccharide capsule of a bacterial cell. The polymeric capsule possesses electrochemical features to sense the electrolyte environment and regulate its compression resistance. We will derive the compressive stress–strain relation and investigate the effect of mechanical–electrochemical coupling mechanisms.

Consider a cylindrical and freely swelling capsule placed in an electrolyte solution and subjected to uniaxial compression in the vertical direction by a permeable platen, as shown in Fig. 7(a). In this case, the capsule can maintain chemical equilibrium with the aid of the external physiological salt solution. Because the double electric layer primarily influences the interfacial interaction and has little effect on the swelling of a bacterial cell, we assume that the electric potential in the freely swelling capsule is homogeneous. Thus, its Cauchy stress is expressed as

$$\boldsymbol{\sigma} = \frac{Nk_B T}{J} (\mathbf{F} \cdot \mathbf{F}^T - \mathbf{I}) - \Lambda \mathbf{I}. \quad (21)$$

The compressive stress between the permeable platen and the capsule contains two parts,  $\sigma_c = \sigma_1 + p$ , where  $\sigma_1$  is the Cauchy stress in the vertical direction and  $p$  is the fluid pressure, which is generally small and can be neglected. Using Eq. (21), we have

$$\sigma_c = \frac{Nk_B T}{\lambda_1 \lambda_2 \lambda_3} (\lambda_1^2 - 1) - \Pi, \quad (22)$$

1 where the stretch ratios  $\lambda_i (i=1,2,3)$  in the three principal directions of the capsule  
 2 are

$$3 \quad \lambda_1 = \lambda_0 (1 + \varepsilon_c), \quad \lambda_2 = \lambda_3 = \lambda, \quad (23)$$

4 respectively, where  $\varepsilon_c$  is the normal compressive strain,  $\lambda_0$  is the stretch ratio of the  
 5 freely swelling capsule, and  $\lambda$  is the stretch ratio in the lateral direction. To derive the  
 6 compressive stress–strain relation, we first need to determine the two stretching ratios,  
 7  $\lambda_0$  and  $\lambda$ . Only their analytical expressions are given below, and the detailed  
 8 derivations are given in [Appendix II](#).

9 First, the stretch ratio  $\lambda_0$  in the free swelling capsule is homogeneous and  
 10 expressed as

$$11 \quad Nk_B T (\lambda_0^{-1} - \lambda_0^{-3}) = -\frac{k_B T}{v_w} \left[ \ln(1 - \lambda_0^{-3}) + \lambda_0^{-3} + \zeta \lambda_0^{-6} \right] \\
 12 \quad + k_B T \left[ \sqrt{\left( \frac{z_0 C_0}{\lambda_0^3 - 1} \right)^2 + 4c_0^2} - 2c_0 \right], \quad (24)$$

12 where  $c_0$  is the ionic concentration in the external solution.

13 The stretch ratio in the lateral direction,  $\lambda$ , can be determined from the  
 14 equilibrium condition of Cauchy stress  $\sigma_2 = -p$ . It is given by

$$15 \quad \frac{Nk_B T}{\lambda_1 \lambda^2} (\lambda^2 - 1) = \Pi. \quad (25)$$

16 In the case of uniaxial compression, the osmotic pressure is obtained from Eqs.  
 17 (A.21) and (A.23) as

$$18 \quad \Pi = -\frac{k_B T}{v_w} \left[ \ln\left(\frac{J-1}{J}\right) + \frac{1}{J} + \frac{\zeta}{J^2} \right] + k_B T \left[ \sqrt{\left( \frac{z_0 C_0}{J-1} \right)^2 + 4c_0^2} - 2c_0 \right], \quad (26)$$

19 where  $J = \lambda_1 \lambda^2$ . Thus, we can determine the stretch ratio  $\lambda$  from Eqs. (25) and (26).

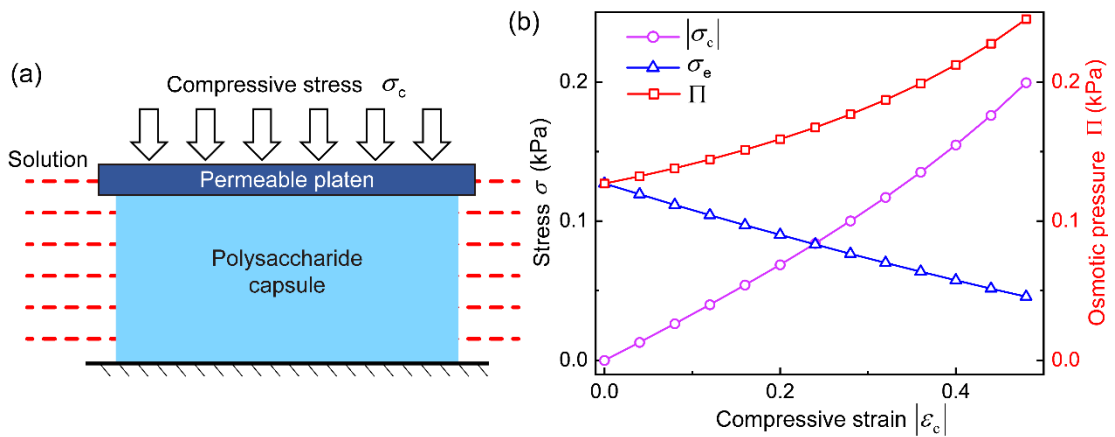
20 Combining Eq. (24) with (26), the compressive stress  $\sigma_c$  is related to the stretch

ratio  $\lambda_1$  by

$$\sigma_c = \frac{Nk_B T}{\lambda_1 \lambda^2} (\lambda_1^2 - 1) + \frac{k_B T}{v_w} \left[ \ln \left( 1 - \frac{1}{\lambda_1 \lambda^2} \right) + \frac{1}{\lambda_1 \lambda^2} + \frac{\zeta}{\lambda_1^2 \lambda^4} \right] - k_B T \left[ \sqrt{\left( \frac{z_0 C_0}{\lambda_1 \lambda^2 - 1} \right)^2 + 4c_0^2} - 2c_0 \right], \quad (27)$$

where the first term is the stress  $\sigma_c$  induced by the deformation of the polymeric network.

For convenience, let  $|\sigma_c|$  and  $|\varepsilon_c|$  denote the magnitudes of the applied compressive stress and strain, respectively. With the swelling of the capsule, the PMs are stretched. As the compressive strain  $|\varepsilon_c|$  increases, the swelling-induced deformation in the vertical direction is offset, and the polymeric stress  $\sigma_c$  decreases, as shown in Fig. 7(b). However, the PMs in the capsule attract water molecules through osmosis, generating larger polysaccharide–water aggregates, which will be trapped by the PM network. The trapped water endows the capsule with high compression resistance. Thus, the osmotic pressure  $\Pi$  primarily contributes to the compression resistance of the capsule, as shown in Fig. 7(b).



**Fig. 7.** Uniaxial compression of a polysaccharide capsule in an external solution. (a) Schematic diagram of uniaxial compression. (b) Theoretical results for the variations in

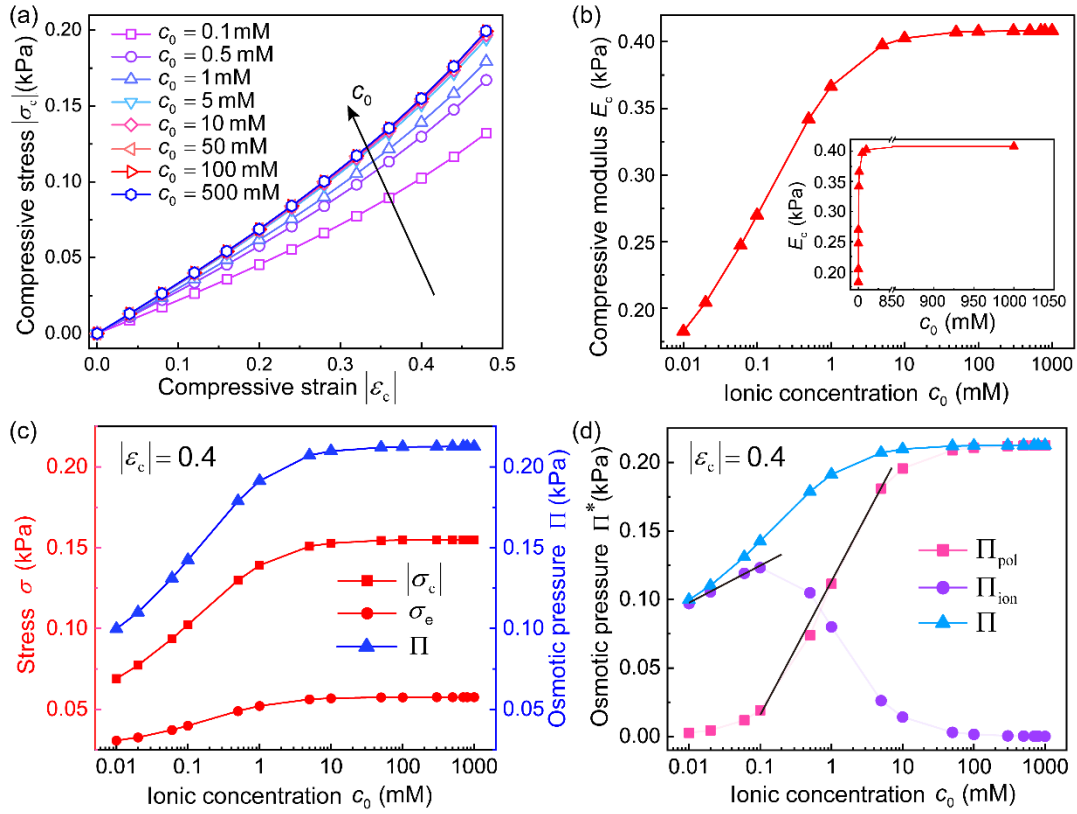


1 the compressive stress  $|\sigma_c|$ , polymeric stress  $\sigma_c$ , and osmotic pressure  $\Pi$  with the  
 2 increase in the compressive strain  $|\varepsilon_c|$ , which are calculated from Eq. (27).

3

4 As can be seen from the compressive stress–strain relation of the capsule in Fig.  
 5 8(a), the stress  $|\sigma_c|$  is approximately linear with the strain  $|\varepsilon_c|$  under a given ionic  
 6 concentration  $c_0$ . The slope of the stress–strain relation is defined as an equivalent  
 7 compressive modulus  $E_c = \Delta\sigma_c / \Delta\varepsilon_c$  to characterize the compression resistance. The  
 8 higher the ionic concentration  $c_0$ , the larger  $E_c$ . Furthermore, Fig. 8(b) shows that  
 9  $E_c$  first increases approximately linearly with  $\log(c_0)$  when  $c_0$  is relatively small  
 10 and finally it approaches a constant of about 0.4 kPa, when  $c_0 > 10$  mM. The theory  
 11 of polysaccharide capsule in Section 2 can be used to reveal the key factors that govern  
 12 the compression resistance. In the capsule consisting of the polymer macromolecule  
 13 network embedded in the solution, the free energy is a function of the molar  
 14 concentration of the solvent, as given in Eqs. (9) and (10). Our theory predicts that  
 15 under a specified strain of  $|\varepsilon_c| = 0.4$ , for example, both the compressive stress  $|\sigma_c|$   
 16 and the osmotic pressure  $\Pi$  increase with the increase in the ionic concentration  $c_0$ ,  
 17 while the polymeric stress  $\sigma_c$  is much smaller than  $\Pi$  and thus negligible, as shown  
 18 in Fig. 8(c). Therefore, the osmosis mainly dictates the compression resistance,  
 19 originating from two parts: the mixing of PMs and the solvent, and the ionic misbalance  
 20 in the capsule and the external solution, as expressed in Eqs. (A.19) and (A.20). The  
 21 theoretical results in Fig. 8(d) demonstrate that the ionic osmotic pressure  $\Pi_{\text{ion}}$  is  
 22 dominant in the osmosis and increases linearly with  $\log(c_0)$  when the ionic  
 23 concentration  $c_0$  is lower than 1.0 mM, while the polymeric osmotic pressure  $\Pi_{\text{pol}}$

1 is dominant in the osmosis and follows a similar rule when  $c_0 > 1.0$  mM. Both the ionic  
 2 and polymeric osmotic pressures approach constant when  $c_0 > 10$  mM. The above  
 3 analysis demonstrates the significance of mechanical–electrochemical coupling  
 4 mechanisms in the mechanical resistance of the polysaccharide capsules of bacterial  
 5 cells.  
 6



7  
 8 **Fig. 8.** Theoretical prediction on the compression resistance of a capsule. (a)  
 9 Compressive stress–strain curves and (b) compressive moduli  $E_c$  under different  
 10 values of the ionic concentration  $c_0$ . (c) The variations in the compressive stress  $|\sigma_c|$ ,  
 11 polymeric stress  $\sigma_e$ , and osmotic pressure  $\Pi$  with respect to the ionic concentration  
 12  $c_0$ . (d) The variations in the osmotic pressures, including the total osmotic pressure  
 13  $\Pi$ , the ionic osmotic pressure  $\Pi_{ion}$ , and the polymeric osmotic pressure  $\Pi_{pol}$ , with

respect to the ionic concentration  $c_0$ . The black lines depict that the osmotic pressures increase linearly with  $\log(c_0)$ .

## 5. Conclusions

In summary, we have established a mechanical–electrochemical model to investigate the deformation behaviors of bacterial cells. By comparing it with our experiments, we demonstrate that this theory can well interpret the coupled mechanical and electrochemical mechanisms underlying the three-stage nanoindentation response of wild-type *K. pneumoniae* cells under different electrolyte conditions. It is revealed that the electrostatic double-layer force between the bacterium and the AFM tip, which depends on the electrolyte concentration, plays a significant role in the first stage of nanoindentation, while the mechanical and electrochemical properties of the polysaccharide capsule and the mechanics of the cell envelope dominate the second and third stages. Furthermore, we use the proposed model to examine the regulatory role of electrolyte concentration on the stress–strain relation of the polysaccharide capsule of a bacterial cell under uniaxial compression. The results show that the present theory can also be applied to investigate the mechanical and electrochemical buffering effect of the capsule on the osmotic adaptation and to characterize the mechanical properties of diverse bacterial cells. The results obtained from this work can deepen our understanding of the mechanical–electrochemical properties of bacterial cells, and the developed analytical approach can be used to study some bacteria-related medical issues.

## **CRediT authorship contribution statement**

**Huanxin Zhang:** Established the theoretical model and performed the numerical simulations. **Huabin Wang:** Performed the nanoindentation experiments. **Yuan Gao:** Performed the compression resistance analysis. **Kaixuan Zhang:** Performed the analysis of the experimental data. **Dominic Vella:** Participated in the theoretical analysis and discussion. **Xi-Qiao Feng:** Conceived the problem and designed the study. All authors wrote the paper.

## **Declaration of competing interest**

The authors declare that they have no known competing financial interests or personal relationships that could have appeared to influence the work reported in this paper.

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## Appendix I. Implementation of the mechanical–electrochemical model into FEM

To demonstrate the present theory via numerical simulations, we implement the proposed theory into the commercial software *Abaqus 6.14*. Eq. (15) shows the stress field in the capsule depends on three fields,  $\mathbf{F}$ ,  $C_\alpha(\mathbf{X})$ , and  $\hat{\mathbf{D}}(\mathbf{X})$ . For illustration, we only consider two species of ions with concentrations  $C_+$  and  $C_-$  in the external solution and the capsule. To easily implement the theory into FEM, we perform a Legendre transformation to transform the free energy function  $w$  into  $\hat{w}$  (Hong et al., 2009; Marcombe et al., 2010; Toh et al., 2014):

$$\hat{w} = w - (\mu_+ - e\Phi)C_+ - (\mu_- + e\Phi)C_- - \mu_w C_w - \hat{\mathbf{E}}^T \cdot \hat{\mathbf{D}}. \quad (\text{A.1})$$

Here, the condition of molecular incompressibility is guaranteed by substituting Eq. (13) into the Helmholtz free energy function density Eq. (7), without introducing the Lagrange multiplier  $\Lambda$ .

Let  $c_0$  denote the true concentration of the ions in the external solution. Using Eq. (17), the chemical potential in the external solution can be given by

$$\mu_+ = e\Phi_e + k_B T \ln \left( \frac{c_0}{c_+^{\text{ref}}} \right), \quad (\text{A.2})$$

$$\mu_- = -e\Phi_e + k_B T \ln \left( \frac{c_0}{c_-^{\text{ref}}} \right), \quad (\text{A.3})$$

where  $\Phi_e$  is the electric potential in the external solution.

The ionic equilibrium condition requires that the electrochemical potential in the capsule is equal to that in the external solution. Thus, we have

$$e(\Phi - \Phi_e) + k_B T \ln \left[ \frac{C_+}{(J-1)c_0} \right] = 0, \quad (\text{A.4})$$

$$e(\Phi - \Phi_e) + k_B T \ln \left[ \frac{(J-1)c_0}{C_-} \right] = 0. \quad (\text{A.5})$$

Then, the condition of ionic equilibrium is written as

$$\frac{(J-1)c_0}{C_-} = \frac{C_+}{(J-1)c_0}. \quad (\text{A.6})$$

Because the double electric layer has less effect on the swelling of a bacterial cell, we assume that the electric potential in the polysaccharide capsule is uniform,  $\partial\Phi/\partial\mathbf{X}=0$ , and that the capsule is electroneutral. Thus, Eqs. (1) and (A.1) reduce to

$$z_0 C_0 - C_- + C_+ = 0, \quad (\text{A.7})$$

$$\hat{w} = w_n + w_s + w_i - k_B T C_+ \ln\left(\frac{C_+}{\nu_w C_w c_+^{\text{ref}}}\right) - k_B T C_- \ln\left(\frac{C_-}{\nu_w C_w c_-^{\text{ref}}}\right) - \mu_w C_w. \quad (\text{A.8})$$

Then  $C_+$  and  $C_-$  can be solved from Eqs. (A.6) and (A.7) as

$$C_- = \frac{1}{2} \left( \sqrt{4c_0^2 (J-1)^2 + (z_0 C_0)^2} + z_0 C_0 \right), \quad (\text{A.9})$$

$$C_+ = \frac{1}{2} \left( \sqrt{4c_0^2 (J-1)^2 + (z_0 C_0)^2} - z_0 C_0 \right). \quad (\text{A.10})$$

Consequently,  $\hat{w}$  can be expressed as a function of the deformation gradient, the ionic concentration in the external solution, and the fixed charge density on the capsule, i.e.,  $\hat{w}(\mathbf{F}, c_0, C_0)$ . When a bacterial cell immersed in a solution reaches the state of electrochemical equilibrium, the ionic concentration in the external solution and the charges fixed in the capsule change little with the swelling of the capsule. Substituting the free energy function  $\hat{w}(\mathbf{F}, c_0, C_0)$  into (3) gives

$$\boldsymbol{\sigma} = J^{-1} \mathbf{F} \cdot \frac{\partial \hat{w}(\mathbf{F}, c_0, C_0)}{\partial \mathbf{F}}. \quad (\text{A.11})$$

The constitutive equation in Eq. (A.11) takes the same form as that of a hyperelastic solid, and the free energy density  $\hat{w}(\mathbf{F}, c_0, C_0)$  depends mainly on the deformation gradient when the electrolyte concentration is fixed. Therefore, the above theory can be readily implemented in the finite element software, *Abaqus*, by coding the free energy density  $\hat{w}$  into the user-defined subroutine, UHYPER.

We first choose a reference state at which the ionic concentration  $c_0$  in the capsule is in an equilibrium state and no mechanical load is applied (Hong et al., 2009).

1 Let  $\lambda_0$  denote the isotropic and free-swelling stretch ratio of the PM network, which  
 2 obeys the relation  $\partial\hat{w}/\partial\mathbf{F}=0$ . The free-swelling state is characterized by the  
 3 deformation gradient,  $\mathbf{F}_0 = \text{diag}(\lambda_0, \lambda_0, \lambda_0)$ . During loading, the deformation gradient  
 4  $\mathbf{F}$  can be decomposed as

$$5 \quad \mathbf{F} = \mathbf{F}_e \cdot \mathbf{F}_0, \quad (\text{A.12})$$

6 where  $\mathbf{F}_e$  is the deformation gradient in the current configuration with respect to the  
 7 free-swelling state. Then the free-energy density in Eq. (A. 8) can be rewritten as

$$8 \quad \hat{w}_e(\mathbf{F}_e, c_0, C_0) = \lambda_0^{-3} \hat{w}(\mathbf{F}, c_0, C_0). \quad (\text{A.13})$$

9 In addition, it is worth mentioning that the ionic concentration in the external  
 10 solution can be mimicked by a temperature-like variable.

11

## 12 **Appendix II. Stretch ratio in a free-swelling capsule**

13 In the considered problem of the capsule under compression, both the fluid phase  
 14 in the external solution and the capsule of the bacterial cell contain water and free ions.  
 15 For simplicity, we consider two species of free ions, which carry positive charges and  
 16 negative charges, respectively. When the system has reached the equilibrium state, the  
 17 fluid phase uniformly distributed in the capsule, and we have  $\sigma_0 = -p$ . From Eq. (25),  
 18 we give

$$19 \quad \Lambda = Nk_B T (\lambda_0^{-1} - \lambda_0^{-3}) + p. \quad (\text{A.14})$$

20 The electrochemical potentials of water and solute in the external solution can be  
 21 written as (Xue et al., 2017)

$$22 \quad \mu_w^{\text{ex}} = v_w \left( p - k_B T \sum_m \bar{c}_m \right), \quad (\text{A.15})$$

$$23 \quad \mu_m^{\text{ex}} = k_B T \ln \frac{\bar{c}_m}{c_m^{\text{ref}}}, \quad (\text{A.16})$$

24 where  $\bar{c}_m$  is the concentration of free ions  $m$  in the external solution.  $m$  can be either

1 “+” or “-”, representing the positive and negative free ions, respectively.

2 At a quasi-equilibrium state, the electrochemical potentials of fluid phases in the  
3 capsule and the external solution are uniform, that is,  $\mu_w = \mu_w^{\text{ex}}$  and  $\mu_m = \mu_m^{\text{ex}}$ . From  
4 Eqs. (16) and (17),  $\Lambda$  and  $C_m$  are derived as

$$\begin{aligned} \Lambda = & -\frac{k_B T}{v_w} \left[ \ln \left( \frac{v_w C_w}{1 + v_w C_w} \right) + \frac{1}{1 + v_w C_w} + \frac{\zeta}{(1 + v_w C_w)^2} \right] \\ & + k_B T \sum_m \left( \frac{C_m}{v_w C_w} - \bar{c}_m \right) + p, \end{aligned} \quad (\text{A.17})$$

$$C_m = v_w C_w \bar{c}_m \exp \left( -\frac{e \Phi z_m + \Lambda v_m}{k_B T} \right). \quad (\text{A.18})$$

7 According to Eq. (A.14), the osmosis of the capsule affects the contractility of the  
8 PM network, and the osmotic pressure is expressed as  $\Pi = N k_B T (\lambda_0^{-1} - \lambda_0^{-3})$ . It can be  
9 divided into two parts: the polymeric osmotic pressure  $\Pi_{\text{pol}}$  arising from the mixing  
10 of PMs and water, and the ionic osmotic pressure  $\Pi_{\text{ion}}$  due to the misbalance of ions  
11 in the capsule and the external solution. They are written as

$$\Pi_{\text{pol}} = -\frac{k_B T}{v_w} \left[ \ln \left( \frac{v_w C_w}{1 + v_w C_w} \right) + \frac{1}{1 + v_w C_w} + \frac{\zeta}{(1 + v_w C_w)^2} \right], \quad (\text{A.19})$$

$$\Pi_{\text{ion}} = k_B T \sum_m \left( \frac{C_m}{v_w C_w} - \bar{c}_m \right). \quad (\text{A.20})$$

14 Since the concentration of the solutes is small, we ignore the contribution of the  
15 solutes to the incompressibility condition in Eq. (13), i.e.,  $1 + v_w C_w = J$ . Thus, the  
16 polymeric osmotic pressure is given by

$$\Pi_{\text{pol}} = -\frac{k_B T}{v_w} \left[ \ln \left( \frac{J-1}{J} \right) + \frac{1}{J} + \frac{\zeta}{J^2} \right]. \quad (\text{A.21})$$

18 In the external solution, both the concentrations of negative and positive ions are  
19 set as  $c_0$ , that is,  $\bar{c}_m = c_0$ . In addition, the effect of  $\Lambda v_m$  on the electric potentials of



ions is in general negligible. By using Eq. (A.18),  $C_+C_- = (v_w C_w c_0)^2$  and the electroneutrality  $C_+ - C_- + z_0 C_0 = 0$ , we obtain

$$C_{\pm} = \frac{1}{2} \left[ \mp z_0 C_0 + \sqrt{(z_0 C_0)^2 + 4(v_w C_w c_0)^2} \right]. \quad (\text{A.22})$$

Thus  $\Pi_{\text{ion}}$  is recast as

$$\Pi_{\text{ion}} = k_B T \left[ \sqrt{\left( \frac{z_0 C_0}{J-1} \right)^2 + 4c_0^2} - 2c_0 \right]. \quad (\text{A.23})$$

From Eqs. (A.14), (A.17), (A.21), and (A.23), it is known that the stretch ratio  $\lambda_0$  of the free swelling capsule can be determined from

$$\begin{aligned} Nk_B T (\lambda_0^{-1} - \lambda_0^{-3}) = & -\frac{k_B T}{v_w} \left[ \ln(1 - \lambda_0^{-3}) + \lambda_0^{-3} + \zeta \lambda_0^{-6} \right] \\ & + k_B T \left[ \sqrt{\left( \frac{z_0 C_0}{\lambda_0^3 - 1} \right)^2 + 4c_0^2} - 2c_0 \right]. \end{aligned} \quad (\text{A.24})$$

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